



1 Phenyl siloxane hybrid xerogels: structure and porous texture

2 P. Moriones^{1,2} · J. C. Echeverria^{1,2} · B. Parra Jose³ · J. J. Garrido^{1,2}

3 Received: 17 December 2018 / Revised: 25 March 2019 / Accepted: 26 March 2019
4 © Springer Science+Business Media, LLC, part of Springer Nature 2019

5 Abstract

6 The aim of this research is to investigate the effect of phenyltriethoxysilane (PhTEOS) and tetraethoxysilane (TEOS) molar
7 ratios as silicon precursors on the structure and porous texture of xerogels. We have prepared phenyl-silane hybrid xerogels
8 from mixtures of PhTEOS and TEOS at pH 10 and 333 K, using ethanol as a solvent. Characterization techniques include
9 ²⁹Si NMR, FTIR, XRD, FE-SEM, HRTEM, TGA-DSC, helium density, and gas adsorption (N₂ at 77 K and CO₂ at 273 K).
10 In order to assess the contribution of the quadrupolar moment of N₂ and CO₂ in the adsorption we obtained the adsorption–
11 desorption isotherm of Ar at 87.3 K for the xerogel synthesized from 50% PhTEOS. The morphology of xerogels changed
12 from aggregates of spherical particles for 20% PhTEOS to lamellae for samples obtained with PhTEOS percentages equal
13 or larger than 60%. The incorporation of phenyl groups into the xerogel matrix caused an increase in the spacing bond
14 between silicon atoms and led to an intramolecular reaction and the formation of lamellar domains. Increasing the PhTEOS
15 molar ratio in the mixture of silicon precursors produced hybrid xerogels with lower specific surface area, pore volume and
16 characteristic energy. The similarity between the isotherms of N₂ at 77 K and Ar at 87.3 K indicates that the main retention
17 mechanism is physisorption and that the variation in the surface chemistry with the incorporation of phenyl groups doesn't
18 inhibit the retention of N₂.

19 **Keywords** Sol–gel · Phenyltriethoxysilane · Hybrid xerogels · Porous texture

20 1 Introduction

21 **AQ1** Phenyl-functionalized silica xerogels are used in several
22 fields due to the outstanding properties resulting from inte-
23 grating in a single material the properties of silica with the
24 phenyl group functionality. For example, silica matrices
25

modified with phenyl groups have been used to entrap active
molecules in their pores that confer specific spectral and
dynamic properties of the dyes (Levy et al. 2006; Pardo et al.
2006; Alsolmy et al. 2018). Another interesting application
is fire retardancy using polymeric materials. In a polymer
nanocomposite, organo-silicates can develop a ceramic
superficial layer during the early stages of combustion,
thereby protecting the underlying material by limiting heat
transfer as well as hampering the diffusion of oxygen and
the evacuation of combustible products (Fina et al. 2006).
A tuneable refractive index has been achieved by modulat-
ing the concentration of phenyl groups (Lin et al. 2018). By
optimizing the matrix structure, refractive indexes of hybrid
silica are comparable to systems that employ zirconium as a
refractive index modifier (Jeong and Moon 2005).

Phenyl-silica xerogels have also been applied to separa-
tions of several binary mixtures, including benzene-toluene,
m-xylene-benzene, m-xylene-toluene, water–ethanol, and
carbon tetrachloride-dichloromethane (Moriones et al. 2011).
Due to the enhanced interaction between phenyl groups on the
analyte and the stationary phase, aromatic substances had a
stronger affinity for the hybrid xerogels than the other analytes.

A1 **Electronic supplementary material** The online version of this
A2 article (<https://doi.org/10.1007/s10450-019-00075-9>) contains
A3 supplementary material, which is available to authorized users.

A4 ✉ J. C. Echeverria
A5 jesus.echeverria@unavarra.es

A6 ✉ J. J. Garrido
A7 j.garrido@unavarra.es

A8 ¹ Departamento de Ciencias, Edif. Los Acebos, Universidad
A9 Pública de Navarra, Campus Arrosadía, 31006 Pamplona,
A10 Spain

A11 ² Institute for Advanced Materials, Edif. Jerónimo de Ayanz,
A12 Universidad Pública de Navarra, Campus Arrosadía,
A13 31006 Pamplona, Spain

A14 ³ Instituto Nacional del Carbon, CSIC, Apartado 73,
A15 33080 Oviedo, Spain

Hybrid xerogels have also been used for the separation of hexane isomers and they were able to differentiate between the adsorption of the mono and the di-branched isomers (Fernandes et al. 2019).

Hybrid xerogel films prepared at pH 10 from phenyltriethoxysilane (PhTEOS) and tetraethylsilane (TEOS) mixtures with 30, 40, and 50% PhTEOS in the mixture of their silicon precursors have been used as fiber-optic sensing elements (Echeverria et al. 2017). The films were affixed at the end of optical fibers by the dip-coating technique. The response of each sensing element in the presence of n-hexane decreased with temperature, which denotes an exothermic process and confirms the role of adsorption in the overall performance of the sensing elements. From calibration curves at different temperatures, the isosteric adsorption enthalpies were obtained. The enthalpy change indicated that the adsorbent-adsorbate interaction prevailed at lower relative pressure, whereas condensation of n-hexane on the meso and macropores was the main mechanism at higher relative pressure.

The overall performance of the hybrid material strongly depends on the nature and number of the organic functional groups incorporated into the network of the host structure, as well as the size and shape of the pores (Pardo et al. 2011). The most common way to introduce the phenyl group into the silica network is through the use of a phenyltrialkoxysilane precursor mixed with another tetraalkoxysilane precursor. Depending on the molar ratio of the phenyl precursor and the reaction conditions in the sol-gel process, the phenyl group can either act as a matrix modifier or it can direct the formation of lamella (Shimajima et al. 1998) and smaller species such as cage structures or ladder-like polymers (Loy 2007). The phenyl group has steric effects that retard condensation reactions and induces the formation of cycles such as polyhedral oligosilsesquioxanes (Loy et al. 2000; Loy 2007). The organic groups in hydrolysed species act as blocking agents to shield one side of the monomer from reactions that would promote cyclation. However, as silanols are converted into siloxane bonds, the polysilsesquioxanes may become sufficiently hydrophobic to induce phase separation as a precipitate or resin rather than gelation (Loy 2007).

In this research we have investigated the effect of PhTEOS/tetraethoxysilane (TEOS) molar ratios as silicon precursors on the structure and porous texture of xerogels, with the aim of learning how to control the final properties of inorganic-organic materials derived from PhTEOS and TEOS.

2 Experimental section

2.1 Xerogels synthesis

PhTEOS and TEOS with purities of at least 98% were obtained from the Fluka Company (Switzerland). Absolute

ethanol GR and aqueous ammonia GR for analysis were supplied by Merck (Darmstadt, Germany). Water was of MilliQ quality. All xerogels were synthesized at pH 10, 333 K, and a 1:6:6 precursor:ethanol:water molar ratio. The PhTEOS percentages in the siliceous mixture varied from 0 to 100%. The required amounts of precursors and ethanol were mixed using a magnetic stirrer in a 30-mL glass container. While stirring, the amount of water was added drop-wise, and the pH of the solution was adjusted to ten by adding 2-M $\text{NH}_3(\text{aq})$ with an automated burette (Titrimo 702 SM, Metrohm, Herisau, Switzerland). The sample containers were closed and stirred in an orbital shaker and then kept until gelation. The alcogels were covered with 5 mL of ethanol, allowed to age at room temperature for 7 days, and then dried at 295 ± 2 K under atmospheric pressure.

2.2 Sample characterization

The ^{29}Si cross-polarization magic angle spinning (CPMAS) solid-state NMR spectra were acquired on a Bruker AV-400 MHz spectrometer (Billerica, USA) operated at 79.5 MHz for ^{29}Si . The spectra were recorded at room temperature, with chemical shifts reported in ppm relative to tetramethylsilane (TMS). The sample-rotation frequency was 5 kHz for ^{29}Si . Classical notation was used for the ^{29}Si NMR studies: T for silicon with three bridging oxygen atoms (RTEOS) and Q for silicon with four bridging oxygen atoms (TEOS). T and Q notations are usually completed by an I index (T_i , $i=0, 1, 2$ or 3 ; Q_i , $i=0, 1, 2, 3$ or 4), which represents the number of oxo bridges in Si-O-S bonds. The ^{29}Si spectra were ^1H decoupled.

Fourier-transformed infrared (FTIR) spectra were obtained using a Nicolet Avatar 360 FTIR spectrometer (Madison, USA). For each sample, we recorded 32 scans in the $4000\text{--}400\text{ cm}^{-1}$ spectral range with a resolution of 4 cm^{-1} . The KBr pressed-disc technique was used at two sample concentrations: 2 and 0.6 mg dispersed in 198 and 199.4 mg of KBr, respectively. The first one provides detail in the $4000\text{--}2200\text{ cm}^{-1}$ region where hydroxyl and C-H bonds appear. The second one avoids signal saturation in the $2200\text{--}400\text{ cm}^{-1}$ range and makes it easier to analyse the bonds attributed to siloxanic structure. The discs were heated in a furnace at 423 K overnight to minimise the water adsorbed on KBr and the samples (Madejova and Komadel 2001).

X-ray diffraction (XRD) patterns were acquired at room temperature using a Rigaku D-max instrument (Rigaku, Tokyo, Japan) with a copper rotating anode and a graphite monochromator, which was used to select the $\text{CuK}_{\alpha 1/2}$ wavelength. The device was used at 40 kV and 80 mA. The measurements were carried out in step-scan mode for $5^\circ \leq 2\theta \leq 60^\circ$, in steps of 0.03° with a counting rate of one step s^{-1} .

147 Micrographs were obtained using a field-emission scanning
148 electron microscope (FE-SEM) from Carl Zeiss Ultra
149 Plus (Germany). This apparatus permits the observation of
150 samples without metallic coatings. An InLens detector and a
151 2-kV working voltage were used. High-resolution transmission
152 electron microscopy (HRTEM) images were obtained at
153 200 kV with a JEOL 2000 FXII microscope (0.28 nm point-to-
154 point spatial resolution) equipped with an Oxford Instruments
155 INCA 200 energy dispersive spectrometer.

156 Simultaneous thermogravimetry and differential scanning
157 calorimetry (TGA-DSC) analyses of xerogels were performed
158 using a thermogravimetric analyser SETARAM, Mod. Setsys
159 Evolution 1600 (Caluire, France), under an air atmosphere.
160 Samples were placed in a platinum crucible, and the heating
161 rate was 10 K min⁻¹. A blank corresponding to a trial under
162 the same conditions but without a sample was subtracted from
163 each run.

164 The skeletal density was measured via helium pycnometry
165 Accupyc 1330, Micromeritics (Norcross, Georgia, USA).
166 Nitrogen adsorption at 77 K and CO₂ adsorption at 273 K
167 were measured using an ASAP 2010 volumetric adsorption
168 analyser from Micromeritics (Norcross, Georgia, USA). We
169 obtained the adsorption–desorption isotherm of Ar at 87.3 K
170 (ASAP 2020HD, Micromeritics, Norcross, Georgia, USA)
171 for the xerogel synthesized from 50% PhTEOS to assess the
172 contribution of quadrupolar moment of N₂ and CO₂ to the
173 adsorption. Before adsorption analysis, the samples were
174 outgassed for at least 15 h at 423 K at the degasification
175 port of the adsorption apparatus with a residual vacuum of
176 7 × 10⁻¹ Pa. Specific surface areas were calculated from the
177 N₂ and Ar adsorption data (molecular cross section 0.162 nm²
178 and 0.143 nm², respectively) by the Brunauer–Emmett–Teller
179 (BET) method according to criteria described by Rouquerol
180 et al. (Rouquerol et al. 1999), and by applying the Dubinin-
181 Radushkevich (DR) method to CO₂ adsorption (molecular
182 cross-section 0.187 nm²). Total pore volume (V_t) was deter-
183 mined from the amount adsorbed at p/p^o ~ 1. Micropore vol-
184 ume was deduced by applying the DR method to N₂ (V_{DR(N₂)})
185 and CO₂ (V_{DR(CO₂)}) adsorption data. Mesopore volume (V_{meso})
186 values were obtained by subtracting the amount adsorbed at p/
187 p^o 0.80 and 0.30. Macropore volume (V_{macro}) was obtained by
188 difference between V_t and the amount adsorbed at p/p^o 0.80.
189 Pore volumes were calculated using liquid-state densities for
190 adsorbates of N₂ at 0.808 g cm⁻³ and CO₂ at 1.023 g cm⁻³
191 (Garrido et al. 1987; Cazorla-Amoros et al. 1996).

192 3 Results and discussion

193 3.1 Gelation time

194 Gelation time (t_g) decreased with increasing PhTEOS molar
195 ratio (Fig. S1). The sample prepared by using TEOS as the

196 only silica precursor did not gel after 10 months due to the
197 stability of the colloidal suspensions (Stöber et al. 1968;
198 Celzard and Marêché 2002). On the other hand, hydrolysis
199 and condensation of samples with 90 and 100% PhTEOS
200 afforded precipitates (Loy et al. 2000; Dong et al. 2005). For
201 mixtures PhTEOS/TEOS, t_g decreased according to a loga-
202 rithmic relationship for molar ratios between 10 and 40%.
203 For a PhTEOS concentration higher than 40%, t_g linearly
204 decreased with time. Because gelation time as measured
205 includes hydrolysis, condensation and colloid destabiliza-
206 tion, the results suggest a change in the limiting step for
207 molar ratios of PhTEOS higher than 40%. At pH 10 silanol
208 groups dissociate and the negative charge stabilizes colloids.
209 Therefore, the increase in the content of phenyl groups may
210 favour the gelation of colloids by decreasing the surface
211 charge density.

212 3.2 Structural characterization of xerogels

213 3.2.1 Nuclear magnetic resonance

214 Figure 1 shows the ²⁹Si NMR spectra for the xerogels
215 obtained from 10, 20, 40, 50, 60 and 80% PhTEOS. It
216 also includes the spectrum for one xerogel obtained from
217 TEOS (0% PhTEOS) after adding NH₄F to induce the
218 gelation of colloids. Ternary silicon species in which
219 silicon atoms are bonded to phenyl groups are identified
220 between -60 and -85 ppm, the maxima being located
221 as follows: T², C₆H₅-Si≡(O-Si)₂(OH) at -68.6 ppm and
222 T³, C₆H₅-Si≡(O-Si)₃ at -77.7 ppm. The presence of ter-
223 nary signals proves that the bonds between silicon and car-
224 bon from the phenyl groups remains after hydrolysis and
225 condensation reactions. Quaternary silicon species from
226 TEOS appeared between -85 and -120 ppm, and the
227 maxima were located as follows: Q², Si≡(O-Si)₂(OH)₂ at
228 -92.0 ppm; Q³, Si≡(O-Si)₃(OH) at -101.3 ppm; and Q⁴,
229 Si≡(O-Si)₄ at -110.3 ppm (Vogt and Brown 1963; Brown
230 et al. 1964; Innocenzi et al. 2003; Park et al. 2008; Koller
231 and Ulke 2011; Rios et al. 2011). The smaller chemical shift
232 of ternary silicon atoms in comparison to quaternary silicon
233 proves that the phenyl group increases the partial negative
234 charge of silicon atoms.

235 The relative intensity of the quaternary and ternary bands
236 changed with the content of PhTEOS in the initial mixture of
237 reagents (Figs. 1, 2). For percentages of the hybrid precur-
238 sor equal or lower than 20%, quaternary signals were more
239 intense than ternary signals, and Q³ more intense than Q⁴.
240 For PhTEOS percentages equal or higher than 40%, ternary
241 signals were more intense than quaternary, and T³ predomi-
242 nated over T² signals. On the other hand, Q⁴ became more
243 intense than Q³ as PhTEOS content increased. For 60 and
244 80% PhTEOS percentages, the Q⁴ signal was stronger than
245 Q³.

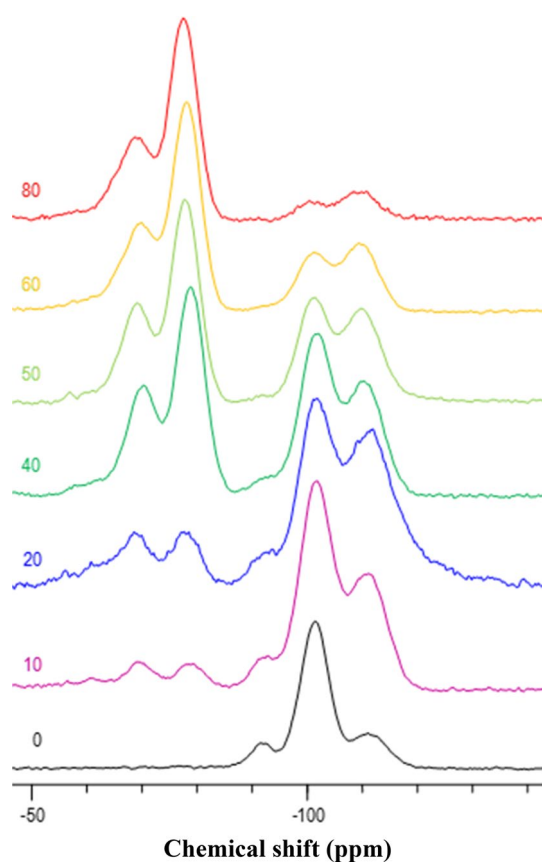


Fig. 1 Solid-state ^{29}Si NMR spectra of xerogels synthesized from different molar percentage of PhTEOS (T^2 , $\text{C}_6\text{H}_5\text{-Si}\equiv(\text{O-Si})_2(\text{OH})$; T^3 , $\text{C}_6\text{H}_5\text{-Si}\equiv(\text{O-Si})_3$; Q^2 , $\text{Si}\equiv(\text{O-Si})_2(\text{OH})_2$; Q^3 , $\text{Si}\equiv(\text{O-Si})_3(\text{OH})$; Q^4 , $\text{Si}\equiv(\text{O-Si})_4$)

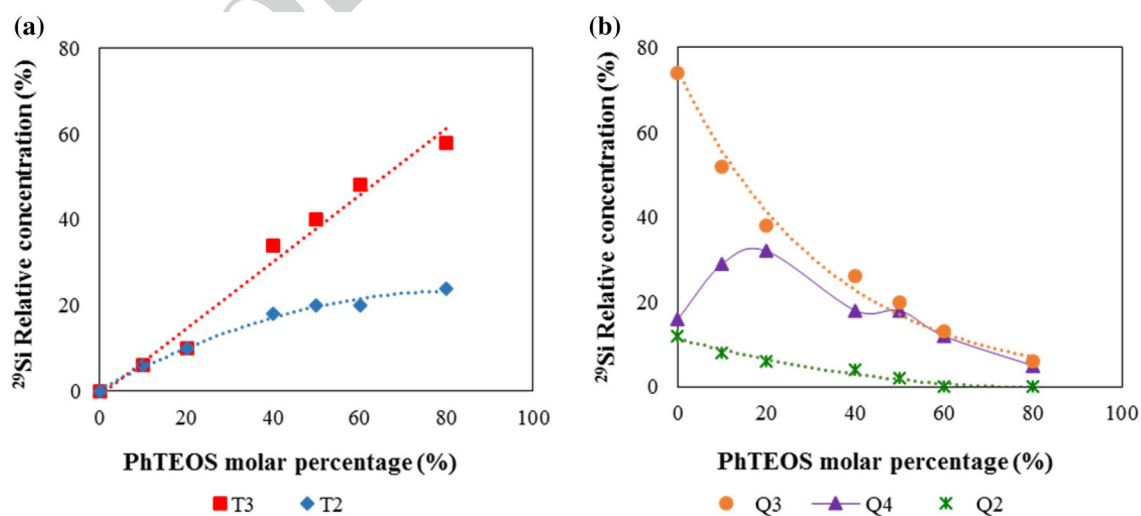


Fig. 2 Relative concentrations of ^{29}Si species in hybrid xerogels as a function of PhTEOS molar percentage: **a** ternary silicon species and **b** quaternary silicon species [T^2 , $\text{C}_6\text{H}_5\text{-Si}\equiv(\text{O-Si})_2(\text{OH})$; T^3 , $\text{C}_6\text{H}_5\text{-Si}\equiv(\text{O-Si})_3$; Q^2 , $\text{Si}\equiv(\text{O-Si})_2(\text{OH})_2$; Q^3 , $\text{Si}\equiv(\text{O-Si})_3(\text{OH})$; Q^4 , $\text{Si}\equiv(\text{O-Si})_4$]

According to Fig. 2a, the relative concentration of T^3 , $\text{C}_6\text{H}_5\text{-Si}\equiv(\text{O-Si})_3$, linearly increased with the PhTEOS content in the initial mixture of precursors, whereas the relative concentration of T^2 , $\text{C}_6\text{H}_5\text{-Si}\equiv(\text{O-Si})_2(\text{OH})$, species stabilized at around 20% for a PhTEOS percentage $\geq 50\%$. The relative concentration of less condensed species Q^2 and Q^3 signals decreased with an increasing PhTEOS content (Fig. 2b). In any case, T^2 and Q^3 were present even for xerogels formed from 80% PhTEOS, which provides evidence that silanol groups are present in the hybrid xerogels.

3.2.2 Infrared spectroscopy

FTIR provides complementary information to discern the bonding of silicon atom and the structure of xerogels. The FTIR spectra of the samples are included in the supplementary information (Fig. S2). The assignment of the bands shown in Table 1 was performed according to current literature (Bellamy 1975; Colthup et al. 1990; Ou and Seddon 1997; Fasce et al. 1999; Fidalgo and Ilharco 2001; Innocenzi 2003; Oubaha et al. 2005; Olejniczak 2005a, b; Al-Oweini and El-Rassy 2009; Li et al. 2009; Qin et al. 2011) and a multiple correlation analysis of the bands having higher intensity. The incorporation of phenyl groups into the structure of hybrid xerogels can be summarized by analysing the FTIR spectra at four wave ranges (Fig. S2): (a) $3800\text{--}2800\text{ cm}^{-1}$, (b) $1800\text{--}1300\text{ cm}^{-1}$, (c) $1300\text{--}900\text{ cm}^{-1}$, and (d) $900\text{--}400\text{ cm}^{-1}$. FTIR spectroscopy confirms the stability of the bond between silicon and a carbon atom from the phenyl group. The presence of the aromatic group resulted in several bands in the $3140\text{--}3010\text{ cm}^{-1}$ range, corresponding to stretching vibration of C-H bonds in the phenyl group (Bellamy 1975; Ou and Seddon 1997; Oubaha

Table 1 Characteristic vibration wavenumbers (cm^{-1}) in FTIR spectra of phenyl functionalized hybrid silica xerogels

Wavenumber (cm^{-1})	Type of vibration	Structural unit
3641	ν OH, H	SiO–H
3427	ν (SiO–H, H)	SiO–H–H
3140	ν (C–H)	C=C–H
3090	ν (C–H)	C=C–H
3078	ν (C–H)	C=C–H
3056	ν (C–H)	C=C–H
3032	ν (C–H)	C=C–H
3018	ν (C–H)	C=C–H
3010	ν (C–H)	C=C–H
2987	ν_{as} (C–H)	–CH ₃
2931	ν_{as} (C–H)	–CH ₂ –R
2906	ν_{s} (C–H)	CH ₃ terminal
1595	ν (C=C)	–C ₆ H ₅
1431	ν (C=C)	–C ₆ H ₅
1234	ν_{as} Si–O–Si (mode LO)	$\equiv\text{Si–O–Si}\equiv$
1134	ν Si–O–Si	$\equiv\text{Si–O–Si}\equiv$ (T ₈)
1097	ν_{as} Si–O–Si (mode TO)	$\equiv\text{Si–O–Si}\equiv$
999	ν_{s} C=C	–C ₆ H ₅
958	ν_{as} (Si–OH)	$\equiv\text{Si–OH}$
800	ν_{s} Si–O	$\equiv\text{Si–O–Si}\equiv$
739	T _{8,y} C–H	–C ₆ H ₅
698	Φ C–H	–C ₆ H ₅
563	ν Si–O	SiO ₂
474	Φ C–H	–C ₆ H ₅
463	δ Si–O–Si	–O–Si–O–

ν stretching vibration; ν_{s} symmetric stretching vibration; ν_{as} antisymmetric stretching vibration; δ , deformation vibration; T₈, in-phase out-of-plane wagging; M, out-of-plane ring deformation vibration; T₈, cubic octamer.

et al. 2005; Li et al. 2009). Specifically, a doublet with the largest intensity at ~ 3074 and 3051 cm^{-1} ; another doublet of low intensity at ~ 3140 and 3090 cm^{-1} , and a triplet of medium intensity at ~ 3039 , 3016 and 3007 cm^{-1} ; which are related to several C–H stretching modes. The bands in the 1800 – 1300 cm^{-1} range are attributed to the stretching vibration of a phenyl framework (Lin et al. 2018). Several vibration modes of phenyl groups bonded to silicon (Si–C) appeared at ~ 999 , 739 , 698 and 474 cm^{-1} (Bellamy 1975; Colthup et al. 1990; Ou and Seddon 1997; Fidalgo and Ilharco 2001; Olejniczak et al. 2005a, b; Fina et al. 2006; Al-Oweini and El-Rassy 2009; Li et al. 2009).

The SiO₂ network changed when increasing PhTEOS content (Innocenzi 2003; Fidalgo and Ilharco 2004) (Fig S3). This figure includes the variation of absorbance as a function of wavenumber, as well as the first and second derivative of the absorbance. As PhTEOS content went up, the 1097 cm^{-1} band and the shoulder at 1234 cm^{-1} ,

which are associated with the siloxane bond in amorphous gels, decreased in intensity, and a narrow band appeared at 1134 cm^{-1} , whose absorbance increased with the concentration of the hybrid precursor. Although the band at 1134 cm^{-1} has been taken as evidence for the presence of strainless cage structures of the T₈–T₁₄ type (Vogt and Brown 1963; Brown et al. 1964; Fasce et al. 1999; Dong et al. 2005; Qin et al. 2011) and highly symmetrical structure with (Si–O)₄ ring subunits (Park et al. 2008), this band is also found in the spectra of liquid PhTEOS used as precursor.

Another relevant effect of increasing the PhTEOS molar ratio is the decrease in intensity of the band with a maximum at around 3414 cm^{-1} , which is attributed to different contributions of geminal, vicinal and H-bonding silanol groups (Colthup et al. 1990; Fidalgo and Ilharco 2001; Innocenzi 2003; Innocenzi et al. 2003; Olejniczak et al. 2005a, b; Oubaha et al. 2005). In any case, isolated silanol groups (Colthup et al. 1990; Fidalgo and Ilharco 2001; Innocenzi et al. 2003; Oubaha et al. 2005) remained in the hybrid xerogels even for PhTEOS molar ratios as high as 80%, which is confirmed by the presence of the band at 3641 cm^{-1} . The presence of these bands backs up the results described in the Sect. 3.2.1 devoted to ²⁹Si NMR.

3.2.3 X-ray diffraction spectroscopy

X-ray diffraction patterns for samples synthesized from 10, 20, 40, 50, 60, and 80% PhTEOS molar percentages are shown in Fig. 3a. All patterns show one broad peak, characteristic of amorphous silica materials (Kamiya et al. 1998; Garcia-Cerda et al. 2002), with a maximum that shifted from $2\theta \sim 22.2^\circ$ for 10% PhTEOS to $2\theta \sim 19.3^\circ$ for 80% PhTEOS. For PhTEOS molar ratios equal or higher than 40%, a distinct peak developed, whose maximum varied from $2\theta \sim 6.3^\circ$ for 40% PhTEOS to $2\theta \sim 7.2^\circ$ for 80% PhTEOS. Figure 3b shows the variation of the d-spacing deduced from the two bands as a function of the molar percentage of the hybrid precursor.

The broad peak between 19.3° and 22.2° is related to the spacing between silicon atoms connected by means of an oxygen bridge (Lana and Seddon 1998). This distance increased from 0.40 nm for the silica xerogel synthesized from 10% PhTEOS to 0.46 nm for 80% PhTEOS. The shift of this band is attributed to steric and inductive effects of the intercalated phenyl groups in the matrix of siloxane bonding, as demonstrated by the T² and T³ signals in the ²⁹Si NMR spectra.

On the other hand, the peak that appeared at $2\theta < 10^\circ$ became sharper and shifted to a wider angle with increasing PhTEOS concentration. Bragg d-spacing decreased from 1.41 nm for 40% PhTEOS to 1.23 nm for 80% PhTEOS (Fig. 3b). The origin of this peak is still controversial. It has been assigned to the presence of fourfold

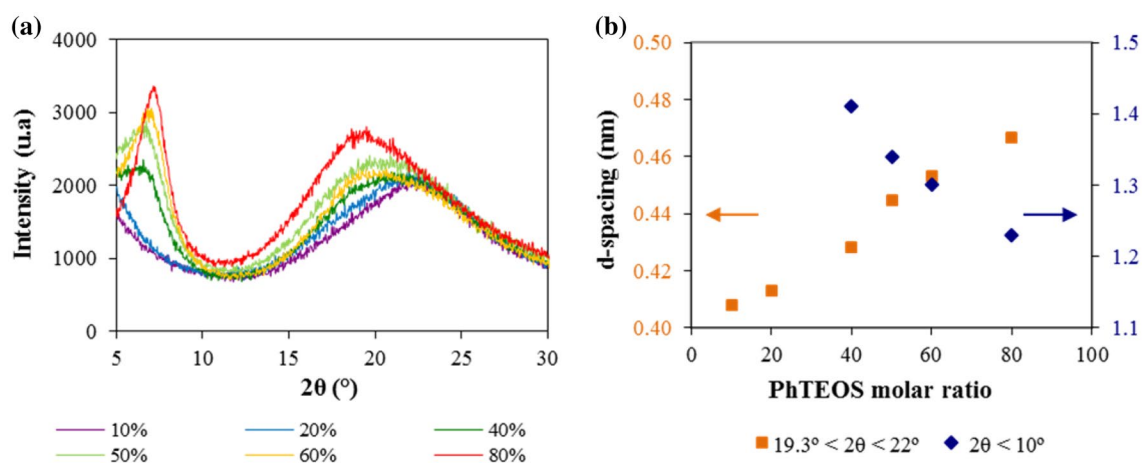


Fig. 3 **a** X-ray diffraction patterns of selected hybrid xerogels, and **b** d-spacing deduced from the two bands as a function of PhTEOS molar percentages

346 siloxane rings (Yoshino et al. 1990; Kamiya et al. 1998;
 347 Fidalgo and Ilharco 2001), but it may originate in a dis-
 348 crete structural unit in the matrix based on an octameric
 349 silicon arrangement (Lana and Seddon 1998; Orel et al.
 350 2005). While tetraalkoxysilanes like TEOS undergo inter-
 351 molecular branching and gelation, enhancing steric hin-
 352 drance—due to a bulky organic group such as phenyl—
 353 strongly favors an intramolecular reaction and cyclics
 354 formation at the expense of intermolecular condensation
 355 (Zhang et al. 2006, 2008; Alauzun et al. 2008; Choi et al.
 356 2011). Gelation would confirm that ordered domains are
 357 embedded in amorphous oligomers (Loy 2007).

358 Another hypothesis is that the structure consists of a
 359 siloxane network and organic layers containing ordered
 360 arrangements. The self-assembly process from co-con-
 361 densation of PhTEOS and TEOS can be summarized in
 362 three imbricated steps. Hydrolysis of the hybrid pre-
 363 cursor results in amphiphilic species that include hydrophilic
 364 silanols and hydrophobic phenyl groups. The amphiphilic
 365 species then assemble and organize themselves into
 366 a solid cross-linked mesostructure through condensation.
 367 The presence of strong van der Waals binding organic
 368 groups drives the self-assembly process (Shimajima
 369 et al. 1998, 2005; Chemtob et al. 2014). The decrease in
 370 d-spacing from 1.41 nm for 40% PhTEOS to 1.23 nm for
 371 80% PhTEOS could be due to a reduction in the thick-
 372 ness of the amorphous silica layer when the molar ratio
 373 of TEOS goes down and the PhTEOS percentage goes up.
 374 The basic aggregates of amphiphilic molecules, such as
 375 the hydrolysis products $C_6H_5Si(OH)_3$, is related to the ratio
 376 of the volume of the molecule to the product of the surface
 377 area and the maximum extension of the non-polar group
 378 (Piazza 2010). For bulky groups, like the phenyl group, the
 379 formation of lamellar structures is favoured.

3.2.4 TEM and FE-SEM micrographs

380

The nanostructures of these hybrid materials were also
 381 studied using TEM (Fig. 4). The micrographs show the
 382 evolution in the nanostructures of the xerogels. The sample
 383 prepared from 20% PhTEOS and 80% TEOS was made
 384 up of aggregates of nanoparticles having around 20 nm
 385 in size, with pores of similar dimensions. The size of the
 386 aggregates increased with the molar percentage of the
 387 hybrid precursor, and the nanostructure changed from
 388 aggregates of nanoparticles to lamellae, which are patent
 389 for 60% and 80% samples. Therefore, the TEM-micro-
 390 graphs confirm the formation of ordered domains shown
 391 in the X-ray diffractograms.

381
382
383
384
385
386
387
388
389
390
391
392

Figure 5 shows FE-SEM images of xerogels at different
 393 molar ratios of PhTEOS: (a) 20%, (b) 40%, (c) 60%, and
 394 (d) 80%. When the PhTEOS content as silicon pre-
 395 cursors increased, the morphology of xerogels changed from
 396 aggregates of spherical particles for 20% PhTEOS to large
 397 lumps of fused spheres where single particles could not be
 398 distinguished for the xerogels obtained with 80% PhTEOS.
 399 Along with the tendency to coalesce, average pore size
 400 and pore size distribution became wider with an increas-
 401 ing molar ratio of the hybrid precursor. Mean particle size
 402 and pore size were estimated by averaging 25 single meas-
 403 urements (“NIS Elements. Advanced solutions for your
 404 imaging world.” 2010). Mean particle size was 42 ± 4 nm
 405 for 20% PhTEOS, 36 ± 11 nm for 40%, 34 ± 3 nm for 60%
 406 and > 1000 nm for 80%. Average pore sizes were 60 ± 10
 407 for 20% PhTEOS, 69 ± 23 nm for 40% nm, 56 ± 7 nm for
 408 60%, and > 500 nm for 80%.

393
394
395
396
397
398
399
400
401
402
403
404
405
406
407
408
409

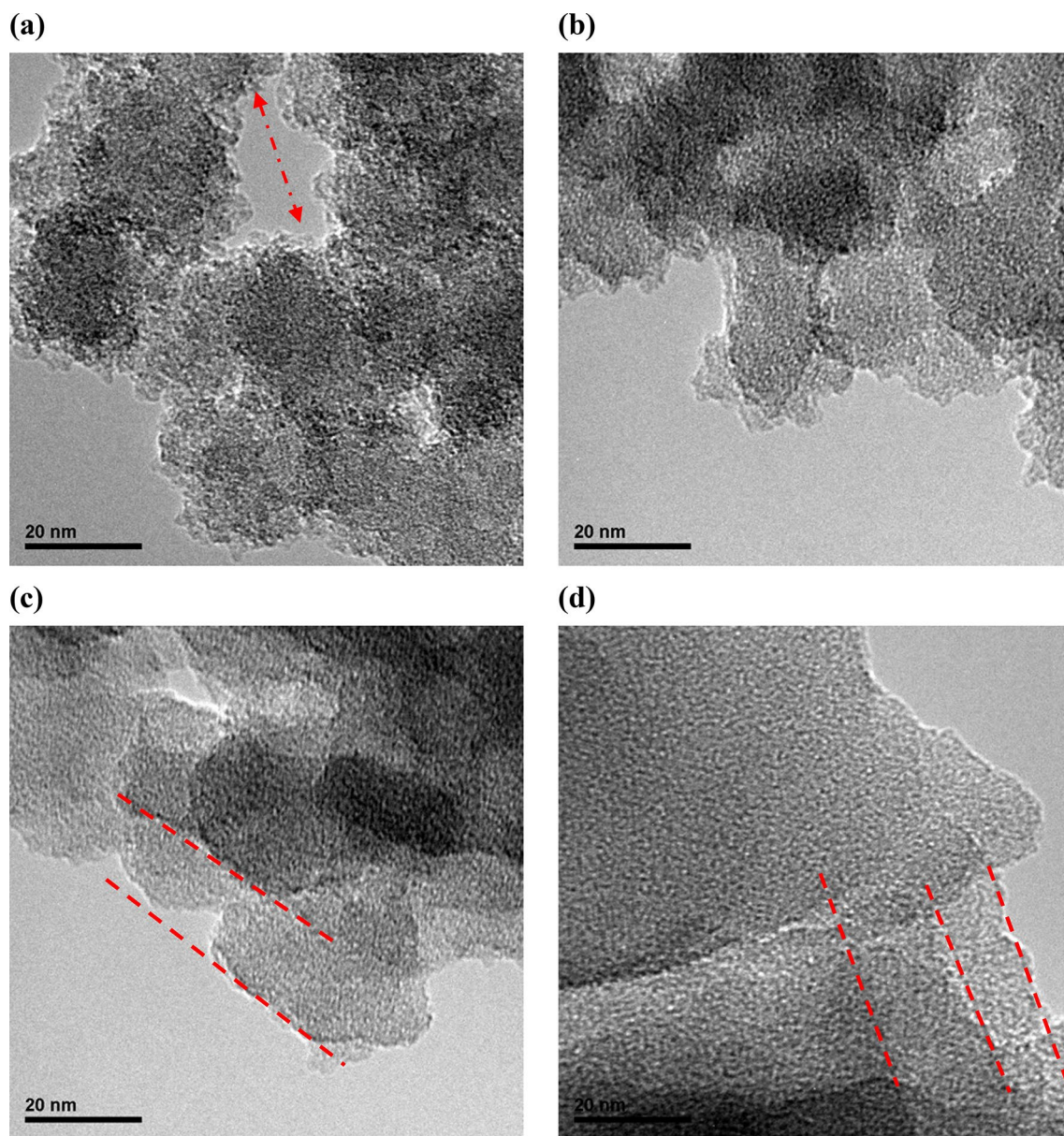


Fig. 4 TEM micrographs of hybrid xerogels: **a** 20% PhTEOS, **b** 40% PhTEOS, **c** 60% PhTEOS, and **d** 80% PhTEOS

3.2.5 Thermogravimetric analysis and differential scanning calorimetry

Figure 6a shows TGA curves of xerogels synthesised from PhTEOS molar ratios ranging from 10 to 80%. Thermograms present three temperature steps which correspond to different processes. The first step goes up to 573 K, and DSC analysis confirms that it is an endothermic process. Mass loss decreased with PhTEOS molar ratio from 7.7% for 10% PhTEOS to 5.1% for 80% (Fig. 6b). This weight loss can be attributed to the removal of physisorbed molecules, in particular, water and ethanol entrapped in the pores. Superficial silanol groups have greater affinity for water and polar

solvents than phenyl groups, and consequently the slope of the curves and the magnitude of the changes in mass in this step suggest that surface polarity decreases with increasing PhTEOS molar ratio. The second step ranges from 573 to 773 K and mass loss varied between 3 and 4%, without significant differences in samples. It can be assigned to the loss of residual ethoxy groups and water produced by condensation of silanol groups on the surface. The third step takes place at a temperature higher than 773 K. The mass loss increased with PhTEOS content from 8.2% for 10% PhTEOS to 40.8% for 80% PhTEOS. This process is exothermic and corresponds to the oxidation of phenyl groups into CO_2 and H_2O .

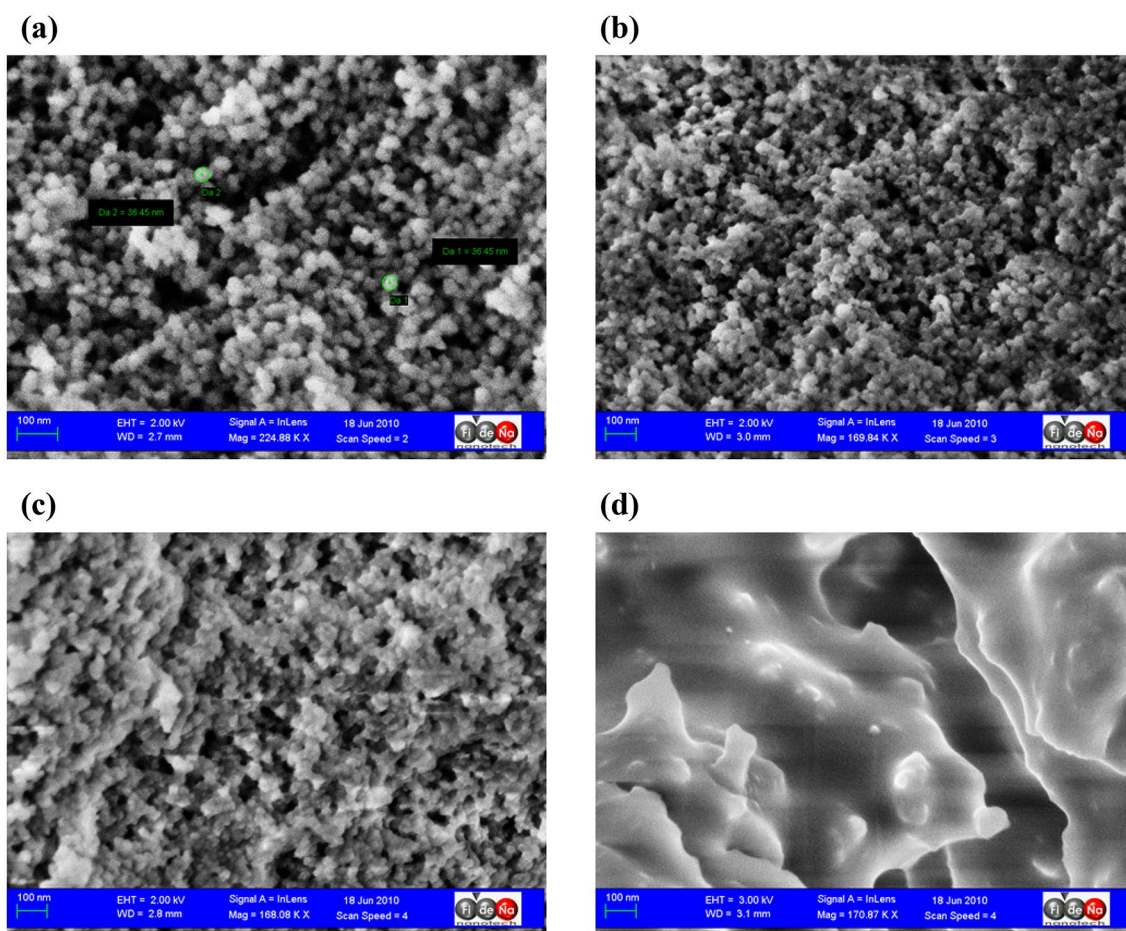


Fig. 5 FE-SEM images of hybrid xerogels synthesized with different percentages of PhTEOS: **a** 20%, **b** 40%, **c** 60% and **d** 80%

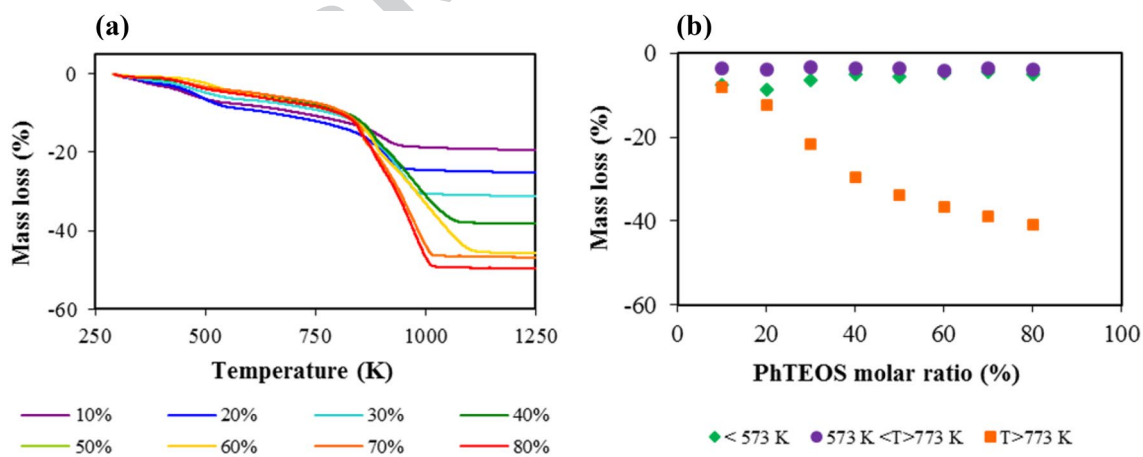


Fig. 6 **a** TGA and **b** The mass loss (%) as a function PhTEOS molar percentage. Heating rate was 10 K min^{-1} under air atmosphere

435 3.3 The porous texture of xerogels

436 The helium density of hybrid xerogels decreased with the
437 amount of PhTEOS and ranged from 1.924 g cm^{-3} for the

xerogel synthesized from 10% PhTEOS to 1.393 g cm^{-3} for
438 the xerogel synthesised from 80% PhTEOS (Fig. S4). The
439 plot of d_{He} as a function of the molar percentage of the hybrid
440 precursors shows that the experimental data can be fitted to
441

442 two linear ranges: between a 10 and 40% PhTEOS molar
 443 ratio the slope of the linear equation is $-1.3 \times 10^{-2} \text{ g cm}^{-3}$
 444 $\%^{-1}$; and between 40 and 80% it is $-3.8 \times 10^{-3} \text{ g cm}^{-3} \%^{-1}$.
 445 The decrease in d_{He} is related to the incorporation of phenyl
 446 groups in the xerogel. The fact that the slope of the variation
 447 of d_{He} was lower in the range 40–80% than between 10 and
 448 40% indicates the contribution of another effect, which could
 449 be the apparition of lamellar domains in the xerogels that
 450 has been discussed in Sect. 3.2.4 devoted to TEM analysis.

451 **3.3.1 Adsorption–desorption isotherms**

452 Figure 7 shows the adsorption–desorption isotherms of the
 453 hybrid xerogels synthesized in the range 10–80% PhTEOS
 454 molar ratios: (a) N_2 at 77 K and (b) CO_2 at 273 K. The insert
 455 shows the relative pressure in a logarithmic scale to detail
 456 the adsorption of N_2 at p/p° below 0.1. The amount of N_2
 457 adsorbed decreased with an increase in content of the hybrid
 458 precursor, and the isotherm knee became sharper. The xerogel
 459 synthesized from 80% PhTEOS scarcely adsorbed N_2
 460 at 77 K. Differences in the adsorbed amount were more
 461 marked for molar ratios higher than 50% and relative pres-
 462 sure below 0.1. Except for 80% PhTEOS, all xerogels pre-
 463 sented N_2 isotherms belonging to Type IV of the IUPAC
 464 classification (Thommes et al. 2015). Type IV isotherms are
 465 given by meso- and macroporous adsorbents. The adsorp-
 466 tion behaviour in meso and macropores is determined by
 467 the adsorbent-adsorptive interactions and by the interactions
 468 between the molecules in the condensed state. In the case of
 469 a Type IVa isotherm, capillary condensation is accompanied
 470 by hysteresis. This occurs when the pore width exceeds a
 471 certain critical width, which is dependent on the adsorption
 472 system and temperature (Thommes et al. 2015). The fact that
 473 the samples adsorbed much more N_2 at relative pressure $p/$
 474 $p^\circ > 0.1$ than at $p/p^\circ < 0.1$ confirms that xerogels are mainly
 475 meso- and macroporous materials.

476 For PhTEOS molar ratios equal or lower than 50%, hys-
 477 teresis loops are type H1, The Type H1 loop is found in
 478 materials which exhibit a narrow range of uniform pores.
 479 Usually, network effects are minimal and the steep, narrow
 480 loop is a clear sign of delayed condensation on the adsorp-
 481 tion branch (Thommes et al. 2015). Xerogels synthesized by
 482 using 60 and 70% PhTEOS molar ratios presented hysteresis
 483 loops Type H2 that have the desorption branch steeper than
 484 the adsorption one. The Type H2(b) loop is also associated
 485 with pore blocking, but the size distribution of neck widths
 486 is now much larger. Examples of this type of hysteresis
 487 loops have been observed with mesocellular silica foams
 488 and certain mesoporous ordered silicas after hydrothermal
 489 treatment.

490 The adsorption of CO_2 also decreased with the molar per-
 491 centage of PhTEOS, but without discontinuity between 50
 492 and 60% as in N_2 isotherms. CO_2 isotherms cover relative

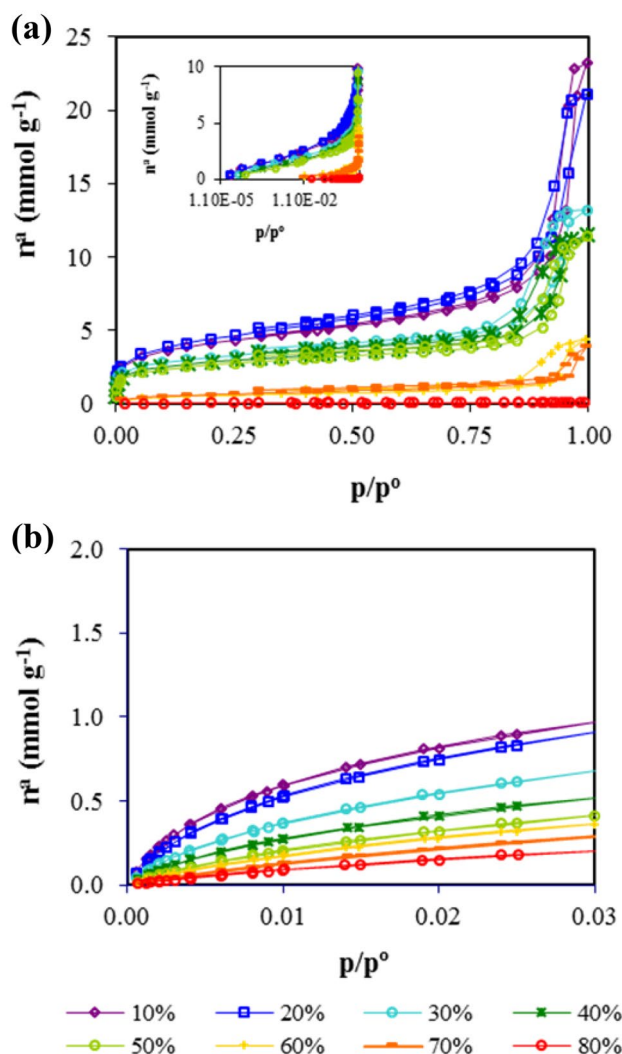


Fig. 7 Adsorption isotherms a N_2 at 77 K b CO_2 at 273 K. The insert plots the relative pressure in logarithmic scale to detail the adsorption at relative pressure below 0.1

pressures up to 0.03, because the CO_2 saturation pressure at 273 K is 34.4 bar.

The adsorption isotherms and the thermal analysis reveal that an increase in the molar percentage of the hybrid precursor brings about a change in the surface chemistry, so xerogels became more hydrophobic, and there is a reduction in the amount of N_2 and CO_2 adsorbed.

3.3.2 Specific surface area

The textural parameters deduced from the N_2 and CO_2 adsorption data are included in Table 2. Correlation analysis of the precursor molar percentages and the textural parameters revealed that increasing PhTEOS in the mixture of silicon precursors produced hybrid xerogels with lower specific surface area, pore volume, and characteristic energy

507 than a xerogel synthesized from TEOS (Table S1). Specific
 508 surface areas deduced by applying the BET method to N₂
 509 adsorption data ($a_{s(BET-N_2)}$) ranged from 340 m² g⁻¹ for 20%
 510 PhTEOS to 4 m² g⁻¹ for 80%. The experimental values con-
 511 firm that the specific area reduction was larger between 50
 512 and 60% PhTEOS. The decrease in the amount of adsorbed
 513 N₂ could be attributed to the weaker cross-linking and the
 514 associated gel flexibility that collapses local domains. At
 515 larger percentages of PhTEOS, it could also be attributed to
 516 the formation of lamellar domains that has been confirmed
 517 by XRD and TEM.

518 In order to assess the contribution of the quadrupolar
 519 moment of N₂ and CO₂ to the adsorption, we obtained the
 520 adsorption–desorption isotherm of Ar at 87.3 K for the
 521 xerogel synthesized from 50% PhTEOS (Fig. S5). Adsorp-
 522 tion–desorption isotherms of N₂ and Ar are Type IV—the
 523 knee, the plateau and the amount adsorbed are similar for
 524 Ar and N₂. The specific surface area was 219 m² g⁻¹ for Ar
 525 and 209 m² g⁻¹ for the $a_{BET(N_2)}$. This similarity indicates that
 526 physisorption is the main retention mechanism, and that the
 527 variation in the surface chemistry with the incorporation of
 528 phenyl groups does not inhibit the retention of N₂.

529 3.3.3 Pore volumes

530 Figure 8 shows the variation of the total pore volume (V_t),
 531 as well as the volume of macro (V_{macro}), meso (V_{meso}) and
 532 micropores (V_{micro}) as a function of the PhTEOS molar per-
 533 centage in the initial mixture of precursors. The total volume
 534 of pores was obtained by converting the amount adsorbed
 535 at the isotherm plateau into a volume of liquid, assuming
 536 that the density of the adsorbate is equal to the bulk liquid
 537 density at saturation (Lowell et al. 2006). The total pore
 538 volume decreased from 0.805 cm³ g⁻¹ for 10% PhTEOS to
 539 0.005 cm³ g⁻¹ for 80% PhTEOS. Comparison of volume of
 540 macro-, meso- and micropores reveals that macropores are
 541 the main contributors to porosity because they account for
 542 more than 60% of the total pore volume (Fig. 8). The volume
 543 of mesopores steadily decreased along with the PhTEOS
 544 percentage and comprised between 10 and 15% of total

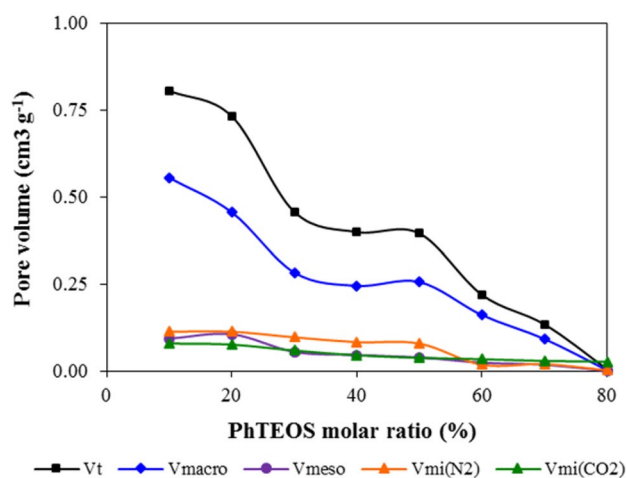


Fig. 8 Pore volume of xerogels as a function of molar percentage of PhTEOS

545 porosity. Both $V_{micro(N_2)}$ and $V_{micro(CO_2)}$ decreased with the
 546 incorporation of phenyl groups in the xerogel. $V_{micro(N_2)}$
 547 was larger than $V_{micro(CO_2)}$ for PhTEOS percentages between 10
 548 and 50%, but for larger percentages of the hybrid precursor
 549 the trend reversed, which reflects the change in the prop-
 550 erties of the surface of xerogels. These results agree with
 551 those described in the sections devoted to FE-SEM images
 552 (Fig. 5), which showed that interparticle volume constitutes
 553 the main contribution to total pore volume and macropore
 554 volume.

555 3.3.4 Characteristic energy

556 For each adsorbate, the characteristic energy decreased with
 557 PhTEOS content, the effect being larger for $E_{c(N_2)}$ than for
 558 $E_{c(CO_2)}$ (Table 2). FTIR spectra demonstrated the incorpora-
 559 tion of phenyl groups into the xerogel and the modification
 560 of siloxane bonds. Phenyl groups on the surface decreased
 561 the surface polarity, which would result in lower characteris-
 562 tic energy. In short, PhTEOS gave rise to xerogels that were
 563 less polar than those obtained from TEOS and, consequently,

Table 2 Textural parameters of phenyl-functionalized hybrid xerogels prepared at pH 10 from mixtures of PhTEOS and TEOS

PhTEOS (%)	Specific area (m ² g ⁻¹)		Pore volume (cm ³ g ⁻¹)				E_c (kJ mol ⁻¹)	
	$a_{BET(N_2)}$	$a_{DR(CO_2)}$	Total	$V_{DR(N_2)}$	$V_{DR(CO_2)}$	Mesopores	N ₂	CO ₂
10	321	145	0.805	0.113	0.079	0.154	16.7	20.8
20	340	144	0.795	0.113	0.075	0.178	16.3	20.3
30	245	117	0.458	0.097	0.059	0.129	16.0	19.4
40	227	94	0.398	0.083	0.046	0.104	15.8	19.1
50	209	85	0.395	0.080	0.039	0.081	14.3	18.2
60	43	80	0.153	0.018	0.036	0.031	10.3	17.7
70	51	78	0.134	0.020	0.034	0.026	10.0	16.6
80	4	61	0.005	0.001	0.026	0.001	8.5	16.0

564 showed lower chemical interaction between adsorbent and
565 adsorbate, and lower the characteristic energy.

566 The correlation coefficient for E_c from N_2 and CO_2 was
567 0.95 (Table S1). When correlation analysis was extended
568 to specific surface area and pore volume, characteristic
569 energy obtained from CO_2 adsorption presented correlation
570 coefficients higher than 0.91 for all textural parameters. On
571 the other hand, E_c from N_2 adsorption only correlated with
572 $a_{s(BET)}$. Therefore, the characteristic energy determined from
573 CO_2 adsorption appears to be a better indicator of textural
574 parameters than $E_{c(N_2)}$.

575 4 Conclusions

ABSTRACT We have investigated the structure and porous texture of
577 phenylsilane hybrid xerogels prepared from mixtures of
578 PhTEOS and TEOS. The gelation time decreased with an
579 increasing PhTEOS molar ratio at pH 10. Increasing the
580 concentration of phenyl groups yielded colloids with a
581 lower density in silanol groups which, in turn, reduced the
582 surface charge density of colloids and favoured gelation.
583 The morphology of xerogels changed from aggregates of
584 spherical particles for 20% PhTEOS to lamellae for sam-
585 ples obtained with PhTEOS percentages equal or larger than
586 60%. An incorporation of phenyl groups into the xerogel
587 matrix caused an increase in the spacing bond between sili-
588 con atom and promoted an intramolecular reaction and the
589 formation of lamellar domains. FTIR spectra demonstrate
590 that the incorporation of phenyl groups reduces the amount
591 of hydroxyl groups. Phenyl-silica xerogels were thermosta-
592 ble up to 773 K. Increasing the PhTEOS molar ratio in the
593 mixture of silicon precursors produced hybrid xerogels with
594 lower specific surface area, pore volume and characteristic
595 energy. The similarity between the isotherms of N_2 at 77 K
596 and Ar at 87.3 K indicates that the main retention mecha-
597 nism is physisorption, and that the variation in the surface
598 chemistry with the incorporation of phenyl groups does not
599 inhibit the retention of N_2 .

600 **Acknowledgements** This work was backed by the “Ministerio de
601 Ciencia y Tecnología” (Grant No. CTQ2009-07993) and by the
602 “Ministerio de Economía, Industria y Competitividad” (Grant No.
603 MAT2016-78155-C2-2-R). Paula Moriones is grateful to the “Depar-
604 tamento de Industria y Tecnología, Comercio y Trabajo” of the Navarre
605 Government for the fellowships granted (Ref. Number 175/01/08 and
606 269/01/08, respectively). The authors thank the “Servicios Científico-
607 Técnico de Investigación” at the University of Cantabria (Spain) for
608 TGA-DSC analysis.

609 References

610 Alauzun, J., Mehdi, A., Mouawia, R., Reyé, C., Corriu, R.: Synthesis
611 of new lamellar materials by self-assembly and coordination

chemistry in the solids. *J. Sol-Gel. Sci. Technol.* **46**, 383–392 (2008) 612

Al-Oweini, R., El-Rassy, H.: Synthesis and characterization by FTIR
spectroscopy of silica aerogels prepared using several $Si(OR)_4$
and $R^+Si(OR^-)_3$ precursors. *J. Mol. Struct.* **919**, 140–145 (2009) 613

Alsolmy, E., Abdelwahab, W.M., Patonay, G.: A comparative study of
fluorescein isothiocyanate-encapsulated silica nanoparticles
prepared in seven different routes for developing fingerprints on
non-porous surfaces. *J. Fluoresc.* **28**, 1049–1058 (2018) 614

Bellamy, L.J.: *The Infra-Red Spectra of Complex Molecules*. Springer,
New York (1975) 615

Brown, J.F., Vogt, L.H., Prescott, P.I.: Preparation and characterization
of lower equilibrated phenylsilsesquioxanes. *J. Am. Chem. Soc.*
86, 1120–1125 (1964) 616

CazorlaAmoros, D., AlcanizMonge, J., LinaresSolano, A.: Characteri-
zation of activated carbon fibers by CO_2 adsorption. *Langmuir* **12**,
2820–2824 (1996) 617

Celzard, A., Marêché, J.F.: Applications of the sol-gel process using
well-tested recipes. *J. Chem. Educ.* **79**, 854–859 (2002) 618

Chemtob, A., Ni, L.L., Croutx-Barghorn, C., Boury, B.: Ordered
hybrids from template-free organosilane self-assembly. *Chem.*
Eur. J. **20**, 1790–1806 (2014) 619

Choi, S.-S., Lee, A., Lee, H., Baek, K.-Y., Choi, D., Hwang, S.: Syn-
thesis and characterization of ladder-like structured polysilsesqui-
oxane with carbazole group. *Macromol. Res.* **19**, 261–265 (2011) 620

Colthup, N.B., Daly, C.M., Wiberley, S.E.: *Introduction to Infrared and*
Raman Spectroscopy. Academic Press, New York (1990) 621

Dong, H.J., Brook, M.A., Brennan, J.D.: A new route to monolithic
methylsilsesquioxanes: gelation behavior of methyltrimethoxysi-
lane and morphology of resulting methylsilsesquioxanes under
one-step and two-step processing. *Chem. Mater.* **17**, 2807–2816
(2005) 622

Echeverria, J.C., Calleja, I., Moriones, P., Garrido, J.J.: Fiber optic
sensors based on hybrid phenyl-silica xerogel films to detect
n-hexane: determination of the isosteric enthalpy of adsorption.
Beilstein J. Nanotech. **8**, 475–484 (2017) 623

Fasce, D.P., Williams, R.J.J., Mechin, F., Pascault, J.P., Llauro, M.F.,
Petiaud, R.: Synthesis and characterization of polyhedral silses-
quioxanes bearing bulky functionalized substituents. *Macromol-*
ecules **32**, 4757–4763 (1999) 624

Fernandes, J., Fernandes, A.C., Echeverria, J.C., Moriones, P., Garrido,
J.J., Pires, J.: Adsorption of gases and vapours in silica based xer-
ogels. *Colloids Surf. Physicochem. Eng. Asp.* **561**, 128–135 (2019) 625

Fidalgo, A., Ilharco, L.M.: The defect structure of sol-gel-derived
silica/polytetrahydrofuran hybrid films by FTIR. *J. Non-Cryst.*
Solids **283**, 144–154 (2001) 626

Fidalgo, A., Ilharco, L.M.: Correlation between physical properties and
structure of silica xerogels. *J. Non-Cryst. Solids* **347**, 128–137
(2004) 627

Fina, A., Tabuani, D., Carniato, F., Frache, A., Boccacali, E., Camino,
G.: Polyhedral oligomeric silsesquioxanes (POSS) thermal deg-
radation. *Thermochim. Acta* **440**, 36–42 (2006) 628

Garcia-Cerda, L.A., Mendoza-Gonzalez, O., Perez-Robles, J.F., Gon-
zalez-Hernandez, J.: Structural characterization and properties
of colloidal silica coatings on copper substrates. *Mater. Lett.* **56**,
450–453 (2002) 629

Garrido, J., Linares-Solano, A., Martin-Martinez, J.M., Molina-Sabio,
M., Rodriguez-Reinoso, F., Torregrosa, R.: Use of N_2 vs CO_2
in the characterization of activated carbons. *Langmuir* **3**, 76–81
(1987) 630

Innocenzi, P.: Infrared spectroscopy of sol-gel derived silica-based
films: a spectra-microstructure overview. *J. Non-Cryst. Solids*
316, 309–319 (2003) 631

Innocenzi, P., Falcaro, P., Grosso, D., Babonneau, F.: Order-disorder
transitions and evolution of silica structure in self-assembled 632

- 678 mesostructured silica films studied through FTIR spectroscopy. *J. Phys. Chem. B* **107**, 4711–4717 (2003)
- 679 Jeong, S., Moon, J.: Fabrication of inorganic-organic hybrid films for
680 optical waveguide. *J. Non-Cryst. Solids* **351**, 3530–3535 (2005)
- 681 Kamiya, K., Dohkai, T., Wada, M., Hashimoto, T., Matsuoka, J., Nasu,
682 H.: X-ray diffraction of silica gels made by sol-gel method under
683 different conditions. *J. Non-Cryst. Solids* **240**, 202–211 (1998)
- 684 Koller, H., Ulke, S.: Microheterogeneity in phenyl group modified
685 inorganic/organic hybrid gels after aerosol drying or slow solvent
686 evaporation. *Solid State Nucl. Magn. Reson.* **39**, 142–150 (2011)
- 687 Lana, S.L.B., Seddon, A.B.: X-ray diffraction studies of sol-gel derived
688 ORMOSILs based on combinations of tetramethoxysilane and
689 trimethoxysilane. *J. Sol-Gel. Sci. Technol.* **13**, 461–466 (1998)
- 690 Levy, D., Pardo, R., Zayat, M.: Photostability of a photochromic naph-
691 thopyran dye in different sol-gel prepared ormosil coatings. *J.*
692 *Sol-Gel. Sci. Technol.* **40**, 365–370 (2006)
- 693 Li, Y.-S., Wang, Y., Ceesay, S.: Vibrational spectra of phenyltriethox-
694 ysilane, phenyltrimethoxysilane and their sol-gels. *Spectrochim.*
695 *Acta A* **71**, 1819–1824 (2009)
- 696 Lin, W.S., Zheng, J.X., Zhuo, J.N., Chen, H.X., Zhang, X.X.: Charac-
697 terization of sol-gel ORMOSIL antireflective coatings from phen-
698 yltriethoxysilane and tetraethoxysilane: microstructure control
699 and application. *Surf. Coat. Tech.* **345**, 177–182 (2018)
- 700 Lowell, S., Shields, J.E., Thomas, M.A., Thommes, M.: Characteriza-
701 tion of Porous Solids and Powders: Surface Area, Pore Size and
702 Density. Springer, Dordrecht (2006)
- 703 Loy, D.A.: Sol-Gel processing of hybrid organic-inorganic materi-
704 als based on polysilsesquioxanes. In: Kicelbick, G. (ed.) *Hybrid*
705 *Materials*, pp. 225–254. Wiley-VCH Verlag, Weinheim (2007)
- 706 Loy, D.A., Baugher, B.M., Baugher, C.R., Schneider, D.A., Rahimian,
707 K.: Substituent effects on the sol-gel chemistry of organotrialkox-
708 ysilanes. *Chem. Mater.* **12**, 3624–3632 (2000)
- 709 Madejova, J., Komadel, P.: Baseline studies of the clay minerals society
710 source clays: infrared methods. *Clays Clay Miner.* **49**, 410–432
711 (2001)
- 712 Moriones, P., Rios, X., Echeverria, J.C., Garrido, J.J., Pires, J., Pinto,
713 M.: Hybrid organic-inorganic phenyl stationary phases for the
714 gas separation of organic binary mixtures. *Colloids Surf. Physico-*
715 *chem. Eng. Asp.* **389**, 69–75 (2011)
- 716 NIS Elements: Advanced Solutions for Your Imaging World. Nikon
717 Corporation, Amstelveen (2010)
- 718 Olejniczak, Z., Leczka, M., Cholewa-Kowalska, K., Wojtach, K.,
719 Rokita, M., Mozgawa, W.: ²⁹Si MAS NMR and FTIR study of
720 inorganic-organic hybrid gels. *J. Mol. Struct.* **744–747**, 465–471
721 (2005a)
- 722 Olejniczak, Z., Cholewa-Kowalska, M.L.K., Wojtach, K., Rokita,
723 M., Mozgawa W.: ²⁹Si MAS NMR and FTIR study of inorganic-
724 organic hybrid gels. *J. Mol. Struct.* **744–747**, 465–471 (2005b)
- 725 Orel, B., Jese, R., Vilcnik, A., Stangar, U.L.: Hydrolysis and solvol-
726 ysis of methyltriethoxysilane catalyzed with HCl or trifluoroacetic
727 acid: IR spectroscopic and surface energy studies. *J. Sol-Gel. Sci.*
728 *Technol.* **34**, 251–265 (2005)
- 729 Ou, D.L., Seddon, A.B.: Near- and mid-infrared spectroscopy of sol-gel
730 derived ormosils: vinyl and phenyl silicates. *J. Non-Cryst. Solids*
731 **210**, 187–203 (1997)
- 732 Oubaha, M., Etienne, P., Calas, S., Sempere, R., Nedelec, J.M.,
733 Moreau, Y.: Spectroscopic characterization of sol-gel organo-
734 siloxane materials synthesized from aliphatic and aromatic alcox-
735 ysilanes. *J. Non-Cryst. Solids* **351**, 2122–2128 (2005)
- 736 Pardo, R., Zayat, M., Levy, D.: Photostability of a photochromic naph-
thopyran dye in different sol-gel prepared ormosil coatings. *J.*
Sol-Gel. Sci. Technol. **40**, 365–370 (2006)
- Pardo, R., Zayat, M., Levy, D.: Photochromic organic-inorganic hybrid
materials. *Chem. Soc. Rev.* **40**, 672–687 (2011)
- Park, E.S., Ro, H.W., Nguyen, C.V., Jaffe, R.L., Yoon, D.Y.: Infrared
spectroscopy study of microstructures of poly(silsesquioxane)s.
Chem. Mater. **20**, 1548–1554 (2008)
- Piazza, R.: *Soft matter. The stuff dreams are made of.* Springer,
Dordrecht (2010)
- Qin, Y., Ren, H., Zhu, F., Zhang, L., Shang, C., Wei, Z., Luo, M.:
Preparation of POSS-based organic-inorganic hybrid mesoporous
materials networks through Schiff base chemistry. *Eur. Polym. J.*
47, 853–860 (2011)
- Rios, X., Moriones, P., Echeverria, J.C., Luquin, A., Laguna, M., Gar-
rido, J.J.: Characterisation of hybrid xerogels synthesised in acid
media using methyltriethoxysilane (MTEOS) and tetraethoxysilane
(TEOS) as precursors. *Adsorption* **17**, 583–593 (2011)
- Rouquerol, F., Rouquerol, J., Sing, K.: *Adsorption by Powders and*
Porous Solids. Academic Press, San Diego (1999)
- Shimajima, A., Sugahara, Y., Kuroda, K.: Synthesis of oriented inor-
ganic-organic nanocomposite films from alkyltrialkoxysilane-
tetraalkoxysilane mixtures. *J. Am. Chem. Soc.* **120**, 4528–4529
(1998)
- Shimajima, A., Liu, Z., Ohsuna, T., Terasaki, O., Kuroda, K.: Self-
assembly of designed oligomeric siloxanes with alkyl chains
into silica-based hybrid mesostructures. *J. Am. Chem. Soc.* **127**,
14108–14116 (2005)
- Stöber, W., Fink, A., Bohn, E.: Controlled growth of monodisperse
silica spheres in micron size range. *J. Colloid Interface Sci.* **26**,
62–69 (1968)
- Thommes, M., Kaneko, K., Neimark, A.V., Olivier, J.P., Rodriguez-
Reinoso, F., Rouquerol, J., Sing, K.S.W.: Physisorption of gases,
with special reference to the evaluation of surface area and pore
size distribution (IUPAC Technical Report). *Pure Appl. Chem.*
87, 1051–1069 (2015)
- Vogt, L.H., Brown, J.F.: Crystalline methylsilsesquioxanes. *Inorg.*
Chem. **2**, 189–192 (1963)
- Yoshino, H., Kamiya, K., Nasu, H.: IR study on the structural evolu-
tion of sol-gel derived SiO₂ gels in the early stage of conversion
to glasses. *J. Non-Cryst. Solids* **126**, 68–78 (1990)
- Zhang, X., Xie, P., Shen, Z., Jiang, J., Zhu, C., Li, H., Zhang, T., Han,
C.C., Wan, L., Yan, S., Zhang, R.: Confined synthesis of a cis-
isotactic ladder polysilsesquioxane by using a π -stacking and
H-bonding superstructure. *Angew. Chem. Int. Ed.* **45**, 3112–3116
(2006)
- Zhang, Z.X., Hao, J.K., Xie, P., Zhang, X.J., Han, C.C., Zhang, R.B.: A
well-defined ladder polyphenylsilsesquioxane (Ph-LPSQ) synthe-
sized via a new three-step approach: monomer self-organization-
lyophilization-surface-confined polycondensation. *Chem. Mater.*
20, 1322–1330 (2008)
- Publisher's Note** Springer Nature remains neutral with regard to
jurisdictional claims in published maps and institutional affiliations.

Journal:	10450
Article:	75

Author Query Form

Please ensure you fill out your response to the queries raised below and return this form along with your corrections

Dear Author

During the process of typesetting your article, the following queries have arisen. Please check your typeset proof carefully against the queries listed below and mark the necessary changes either directly on the proof/online grid or in the 'Author's response' area provided below

Query	Details Required	Author's Response
AQ1	Author: Please check and confirm that the edit made in the title is correct.	
AQ2	Author: Reference Olejniczak et al. (2005) was repeated twice. Since we have changed to Olejniczak et al. (2005a) and Olejniczak et al. (2005b). Kindly check if action is appropriate.	