Understanding the Formation of Al₁₃ and Al₃₀ Polycations to the Development of Microporous Materials based on Al₁₃-and Al₃₀-PILC Montmorillonites: A Review

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ABSTRACT

Hydrolysis of aluminum cations (Al³+), the third most abundant metal in the Earth's crust, is considered relevant in many academic fields, including materials science and chemical engineering. Al^{III} -polycations and their different uses have also been widely studied, as reflected in the extensive literature in that field. This review summarizes some of those studies, from Al³+ hydrolysis to form Al¹₃ ([Al¹₃O₄(OH)₂₄(H₂O)₁₂]³+) and Al₃₀ ([(Al₃₀O₃(OH)₅₆(H₂O)₂₄)]³+) polycations and their specific use as pillaring agents for montmorillonite, which is the most commonly used clay mineral in Aluminum Pillared Interlayered Clays (Al-PILC) synthesis. The experimental conditions published over the years regarding the synthesis of both these Al^{III}-polycations, as well as the conditions employed to synthesize Al-PILC montmorillonite using Al¹₃ and Al₃₀ polycation solutions, are also summarized. This review highlights some of the findings that have made it possible to explain the formation of Al¹₃- and Al₃₀-PILC montmorillonites, and allow us to clearly understand their differences. Finally, the new tendencies in the development of these materials based on Al-PILC and the applications are also highlighted.

Keywords: Aluminum Pillared Interlayered Clays (AI-PILC), Aluminum pillared montmorillonites, Hydrolysis of aluminum, Polynuclear species, Aluminum polycations, Al₁₃ polycation, Al₃₀ polycation, hydroxo polymers, Keggin-type species.

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1. Introduction

Clay minerals are ubiquitous components in soils and sediments that play an important role in several geological and biological events and dominate numerous environmental and geochemical processes (Wen et al., 2019b). The majority of clays are made of stacked tetrahedral and octahedral layers, the arrangement of which allows three main types of clay to be identified. One of these is the smectite group (2:1), also known as the TOT type because these clays contain one octahedral sheet of alumina (O) between two tetrahedral sheets of silica (T); montmorillonite is one example of smectite (**Figure 1**). These TOT clays have a permanent negative surface charge due to isomorphous substitution, (such as, e.g., Al^{III} by Si^{IV} in the T-layer and/or Al^{III} by Mg^{II} in the O-layer), which is compensated by the presence of cations (such as Na⁺ or Ca²⁺), known as "exchangeable cations", "interlayer cations", or "compensator cations", in the interlayer space (Jlassi et al., 2017; Lazaratou et al., 2020; Motalov et al., 2017; Wen et al., 2019b).

Smectites expand when hydrated, but when heated and dehydrated, their layers collapse, thus making the interlayer surface inaccessible for chemical process applications. To solve this problem, researchers have distanced the clay layers, thereby augmenting the pore volume and allowing clays to maintain their porosity during the hydration and dehydration processes (Figure 2a). To achieve that, several studies have proposed the introduction of stable pillars in the interlayer region, thus converting the clay into a pillared interlayered clay (PILC) (Cool and Vansant, 1998). These pillars are introduced into the interlayered space using two main steps (Figure 2b): an intercalation process consisting of switching the "exchangeable cations" for a larger and bulkier polymeric inorganic cation, via a cation-exchange reaction, or a calcination step, which involves converting the intercalated polycation precursors into rigid metal oxide clusters, or pillars, (e.g. Al₂O₃, TiO₂, Fe₂O₃, etc.) by dehydration and dehydroxylation. The resulting materials, in which pillars are strongly bonded to the clay layers, retain the layered structure of clays without collapsing and have a higher basal spacing, pore volume, and surface area (Cool and Vansant, 1998; Gil et al., 2000a; Nunes et al., 2008;

Osorio-Revilla et al., 2006; Vicente et al., 2013; Yuan et al., 2006; Zhu et al., 2018). The size and shape of two-dimensional porous PILC depend on many factors, including the nature and layer charge of the layered hosts, the pillaring materials used, their valence/charge, the pillaring steps, and the reaction conditions (Gil et al., 2000a; Gil et al., 2007; Zhu et al., 2017; Zuo et al., 2012).

The properties of PILC make these solids suitable for applications in several fields, especially for adsorption and catalysis purposes (Cool and Vansant, 1998; Gil et al., 2011a; Wen et al., 2019a). Clay minerals, whether natural or modified, are considered interesting materials because of their environmentally-friendly features, their abundance, and low cost; of these, montmorillonite has been widely used as a host, due to its structural and physicochemical properties, especially its swelling capacity, high internal surface area, and high cation-exchange capacity (Zhu et al., 2018). Inorganic cations with large particle sizes have been used as pillaring agents, with the most common being the cations of following elements Al, Fe, Zr, Cr, and Ti. However, over the years, other cations, including those of Ni, V, Zn, Ga, Co, Cu, Ta, have also been used, as have Al^{III}-containing composites, such as Al/Fe and Al/Co, amongst others, and even mixed oxides supported as Fe-Mo, Cu-Ce, among others (Cañizares et al., 1999; Galeano et al., 2014; Gil et al., 2000a; Muñoz et al., 2017; Wen et al., 2019a; Zhu et al., 2017).

However, among the large family of polycations, aluminum Keggin ions as pillars, with montmorillonite as the host clay, have been the most common pillared clays studied, in part due to their permanent porosity and higher accessible surface area, which allows their potential application as an adsorbent for contaminants and their use in heterogeneous catalysis. Al-pillared smectites have been widely documented in the past few decades (Vicente et al., 2013; Wen et al., 2019a; Zhu et al., 2018). As such, this review provides an overview of studies related to the formation of Al-PILC using Al₁₃ and Al₃₀ as an intercalating solution. To understand this process over the years, the review starts with some initial studies on aluminum hydrolysis from 1952, then explains the formation of Al₁₃ and Al₃₀ polycations, and,

finally, the formation of Al-PILC. The second part of this review shows how researchers have proposed several conditions in these processes to achieve a successful synthesis of PILC using Al₁₃ and Al₃₀, and we conclude with a section concerning future challenges in this field.

2. Aluminum hydrolysis

Aluminum hydrolysis is considered relevant in many academic fields. In general terms, hydrolysis reactions are considered important in chemistry because they determine the behavior of the species formed, including their interactions with other complexing ligands or with solids (Mesmer and Baes, 1990). The splitting of water molecules results in the release of protons, and the reaction of metal ions with the products of this water splitting results in the formation of hydroxy or oxy complexes in solution. Polynuclear species, such as $[M_X(OH)_Y]^{(XZ-Y)+}$, are often formed; the solution contains more than one cation and many ions that are hydrolyzed. This can happen in the presence of several mononuclear species of the type $[M_X(OH)_Y]^{(XZ-Y)+}$, and those with symmetrical constitutions appear to be the most common. Many of the analytical characterization techniques that have been developed are used to study these species (Abeysinghe et al., 2013b; Zhu et al., 2017).

Metal ions form mononuclear species upon hydrolysis in dilute solutions (< 10⁻⁴ mol/dm³) which implies the fast removal of a proton from a hydrating water molecule (Mesmer and Baes, 1990). The aqueous chemistry of aluminum is complex; it hydrolyzes in water to produce a wide variety of solute molecules and solids, such as aluminum oxy-hydroxide clusters, and aluminum hydroxide phases (Casey, 2005; Casey et al., 2005; Oliveri et al., 2016; Wang et al., 2011). Aluminosilicates and clays, for example, are important sources of aluminum in soils (Abeysinghe et al., 2012). In the case of aluminum cations (Al³+), the following mononuclear hydrolysis products are formed: [Al(H₂O)₆]³+, [Al(H₂O)₅(OH)]²+, [Al(OH)₄]⁻ (Mesmer and Baes, 1990). Although the existence of oligomers in water has been suspected for decades, only a few of them have been synthesized and

crystallized (Casey et al., 2005). The number of possible complexes is higher when polynuclear complexes are formed compared to mononuclear complexes (Sillén, 1954). Although the interpretation of the results obtained depends on the methods used in the preparation and identification (Bottero et al., 1980), the aging time, for example, is considered important for hydrolyzed aluminum solutions because, although the initial steps are fast, condensation reactions become slower as the polynuclear cations become larger. At this point, change occurs slowly with time and depends on aluminum cation speciation, as well as counterions and heterospecies (Wood et al., 1990).

Hydroxide-containing molecules can be placed into two separate classes. The first of these comprises a Keggin-type structural arrangement characterized by central metals tetrahedrally coordinated to oxygen atoms (MO₄), encompassed by octahedra with common edges (Casey, 2005; Casey et al., 2005; Wang et al., 2011). These species are commonly obtained in aqueous solutions upon titration with a hard base and are considered to undergo rapid hydrolysis (Corum et al., 2015; Fairley et al., 2012). The second class of oligomers was discovered more recently and exhibits a characteristic core of edge-shared octahedra, which are organized into cubane-like moieties that are linked together (Casey, 2005; Casey et al., 2005). These species result in an acidic solution oxidizing metal or accompanied by chelating ligands, and are considered to undergo slow hydrolysis (Fairley et al., 2012). The Keggin-type species formation process is mostly governed by the ratio [NaOH_T]/[Al_T] (total sodium hydroxide concentration/total aluminum concentration) (Brosset, 1952). In this regard, Sillen's "core + links" theory, which is related to this neutralization ratio ($r = [NaOH]/[AI_T]$), has been used by many authors to interpret their results in an acidic medium in terms of Al^{III} (Bottero et al., 1982). Brosset et al. (1952, 1954), for example, used potentiometric titration methods to study aluminum complexes while limiting the ratio r = [OH]/[AI] to 2.5. The composition of the resulting complexes agreed with the said theory. Similarly, Bottero et al. (1980, 1982) also used the potentiometric titration method, hydrolyzing solutions of aluminum chloride with sodium hydroxide while varying the ratio $r = [OH]/[AI_T]$ in dilute solution ((AI_T) =

 10^{-1} mol/dm³) at a temperature of 25 °C. Using 27 Al NMR spectroscopy, these authors reported the presence of five species, namely three monomers [(Al(H₂O)₆]³+, [Al(H₂O)₅(OH)]²+ and [Al(H₂O)₄(OH)₂]+), a dimer [(Al₂(OH)₂]⁴+), and the three-dimensional polymer Al₁₃, which has spherical symmetry (Bottero et al., 1980), that had previously been reported by Johansson (1960) and Akitt (1972). They also studied the variation in the percentage of aluminum bound in the various species as a function of the ratio r and the pH, as shown in **Figure 3** (Bottero et al., 1980). In 1982, Bottero et al. used small-angle X-ray scattering (SAXS) to investigate the effect of aging as well as particle shape and structure, finding that the colloidal species' particle morphology changes as a function of time. The distribution curve in **Figure 3** is dependent on the preparation method and shows that monomeric forms are dominant in the solution if OH/Al < 1. If the OH/Al value increases to between 1.5 and 2.3, the main species in solution is Al₁₃, and an increase in the degree of hydrolysis in the solution increases polymer formation (Cool and Vansant, 1998; Fu et al., 1991).

Rakotonarivo et al. (1984) studied the adsorption of sodium tetradecylsulfonate and sodium dodecylsulfonate on three aluminum hydroxide gels ($r = [OH]/[AI_T] = 0$, 2, and 2.5) at various pH values. These authors used the relationship between the molecular area of the absorbed ion, Avogadro's number, and adsorption isotherm data to calculate the surface area of these aluminum hydroxide gels at a pH of 6.5 and 7.5 in dilute solution ((AI_T) = 0.1 mol/dm³). The resulting calculated surface areas varied from 500 to 1200 m²/g and were found to decrease with increasing pH and [OH]/[AI_T] ratio. Mesmer and Baes (1990) published a review of the hydrolysis behavior of ions, including aluminum, and Casey et al. (2001) published a chapter which reviewed polynuclear complexes of aluminum. The various polynuclear species formed during the aqueous hydrolysis of aluminum depend on the tendency of the metal to hydrolyze, which in turn is reflected in the first hydrolysis reaction of the metal cation (Furrer et al., 1992; Mesmer and Baes, 1990). In the case of aluminum, which is an amphoteric element, it is known that, in aqueous solution, at least some of it is readily hydrolyzed in the pH range from 3 to 11 to form polynuclear

species referred to as "hydroxo polymers", "hydroxo complexes", or "polycations" (see *Supplementary Material*) (Bottero et al., 1980; Corum et al., 2015; Wood et al., 1990).

The work by Mesmer and Baes (1990) includes the dependence of hydrolysis on the temperature and distribution of AIIII hydrolysis products as a function of pH at Al^{III} concentrations of 0.1 and 10⁻⁵ mol/dm³. These authors showed how the tendency of a few cations, including Al³⁺, to hydrolyze increases with temperature (lower pH). This tendency is more pronounced in the case of Al^{III}, in which a starting temperature of 25 °C was observed at a pH of about 5 or lower. For a trivalent cation, saturation occurs at close to pH = 6 (slightly less than 6 at 100 °C and slightly higher at 25 °C), with the presence of M(OH)₄- being observed at higher pH values (Mesmer and Baes, 1990). Similarly, Brosset (1952) reported that, close to the point of equivalence, the viscosity of the solution reaches a maximum and its opalescence is flatter. Below pH = 3, aluminum exists exclusively as monomeric $[Al(H_2O)_6]^{3+}$, whereas above pH = 11 it is present as the $[AI(OH)_4]^-$ ion (Wood et al., 1990). This means that there are two processes which depend on the pH: one in the acid range and the other in the alkaline range (Brosset, 1952). The total aluminum concentration is important because the system becomes much more complex when it is increased (Wood et al., 1990). Brosset et al. (1952) assumed a primary protolysis with subsequent "protolytic association" for both environments (alkaline and acid). However, protolysis in an alkaline environment could directly result in the [Al(OH)₄]⁻ ion, which contrasts with what occurs in an acidic environment, where the main ions formed are $[AI(H_2O)_5(OH)]^{2+}$ and $[AI(H_2O)_5(OH)_2]^{+}$. These latter ions may subsequently undergo proteolytic association, in which these and Al3+ ions may participate. If the aluminum ion is in an alkaline environment, it is protolyzed instantaneously to [Al(OH)₄]. If there is a sufficient excess of alkali in the solution, all the aluminum in it is immediately obtained as [Al(OH)₄]⁻ and equilibrium prevails, rapidly reaching a constant pH value (Brosset, 1952).

When the pH is lower than 3 in an aqueous solution, the hydrated Al³+ cation is the primary species present. However, when the pH is increased, the monomeric species $[(Al(H_2O)_5OH)]^{2+}$, $[(Al(H_2O)_5(OH)_2)]^+$, and $[(Al(OH)_4)]^-$ condense to produce dimers $[(Al_2(OH)_2)]^{4+}$, trimers $[(Al_3(OH)_4)]^{5+}$, and a longer list of complex polymeric species, such as $[(Al_6(OH)_{15})]^{3+}$, $[(Al_7(OH)_{17})]^{4+}$, $[(AlO_4Al_{12}(OH)_{24}(H_2O)_{12})]^{7+}$, $[(Al_16(OH)_{38})]^{10+}$, $[(Al_30O_8(OH)_{56}(H_2O)_{24})]^{18+}$, $[(Al_9(OH)_n)]^{(27-n)+}$, etc. These include the tridecamer species ϵ -Al $_{13}$ ($[Al_{13}O_4(OH)_{24}(H_2O)_{12}]^{7+}$) (Abeysinghe et al., 2012; Corum et al., 2015; Wu et al., 2016). Only a few of these aluminum-containing oligomers have been isolated, and only a limited amount of structural data are available (Casey, 2005; Wang et al., 2011). A large number of these polymers therefore remain uncharacterized and unidentified, partly due to the fact that 27 Al NMR spectroscopy only detects complexes with a tetrahedral site (Casey, 2005).

Kinetic phenomena in the hydrolysis process are pertinent because they have also been emphasized in some studies (Bottero et al., 1980). Thus, Brosset (1952) concluded that the reaction of aluminum in an alkaline aqueous environment is a first-order reaction, whereas in an acid environment it is a second-order reaction. This is related to the aging process and the formation of α -gel or products, respectively. Similarly, Casey (2005) reported kinetic aqueous interactions of the known aluminum hydroxide polyoxocations. In addition, these authors were pioneers in researching exchange kinetics involving Keggin-type species, using them to research similar reactions such as on extended mineral surfaces.

3. Al₁₃ and Al₃₀ formation

Al₁₃ and Al₃₀ structures are explained in the *Supplementary Material*. Polynuclear aluminum species can form naturally. Thus, Furrer et al. (1992) simulated natural conditions in the laboratory and found that the total aluminum concentration affects the formation of Al₁₃, with high concentrations favoring this formation and lower concentrations shifting formation to higher pH. For example, according to the equilibrium calculations, Al₁₃ formation at an [Al(III)]_T concentration

of 10^{-3} mol/dm³ starts at pH = 4.2, whereas this process starts at a pH of between 5.0 and 6.0 at 10⁻⁵ mol/dm³. At concentrations below 10⁻⁵ mol/dm³, this transformation to Al₁₃ is unlikely. These authors also reported that energetic stirring is not necessary and that more than 67% of the Al monomers were converted into Al₁₃ polycations. The polymer maintains its +7 charge at a pH of less than 6 but loses nearly all its positive charge at a pH of between 6 and 7, after a series of deprotonation steps, which lead to the formation of species such as Al₁₃5+, Al₁₃3+, and Al₁₃+ (Abeysinghe et al., 2013b; Cool and Vansant, 1998; Vaughan, 1988). At a pH above 6, the equilibrium conditions are totally altered and aggregation due to proton loss from Al₁₃ occurs (Mertens et al., 2016). Other complexes are also found in solution as a result of the interaction between dissolved metal cations and the partially neutralized Keggin ions (Furrer et al., 1992). The acidity is important to obtain good control of the charge of the Al₁₃ polyoxycations because the OH/Al ratio is directly related to the pH of the solution (Cool and Vansant, 1998). In this regard, Vaughan published a study of the charge variation due to pH, showing that, as the pH increases, the charge decreases, and considering the possibility that, in solution at pH > 3, polymers with no less than two different charges are likely to be present (Vaughan, 1988).

Nazar et al. (Fu et al., 1991; Nazar et al., 1992) studied the hydrolysis of aluminum in an aqueous solution at high temperatures (80 - 95 °C) and identified three new polyoxoaluminum cations (denoted as AIP₁, AIP₂, and AIP₃) formed by thermal transformation of the tridecamer cation AI₁₃ after aging for a few days. Using kinetic NMR analysis and quantitative NMR studies, these authors proposed that the first step in the thermal evolution involves degradation of the tridecamer cluster (AI₁₃), which loses one octahedral AI unit to form the intermediate species AIP₁ (AI₁₂, a defective AI₁₃ structure). This species contains four active sites and react via condensation/polymerization to form larger and more stable polycationic species. The next step is the dimerization of two AIP₁ species to form the AIP₂ cluster, tentatively formulated as AI₂₄O₇₂, which is transformed into AIP₃ upon further heating, despite being stable in solution. This transformation involves distortion of

the Al-tetrahedral sites (Fu et al., 1991; Nazar et al., 1992) (**Figure 4**). The transformation of Al^{III} into Al₁₃ and then AlP₂ is favored by hydrolysis (OH/Al = 2.25) of AlCl₃ solutions (0.25 mol/dm³) at 80 °C. Thus, Parker et al. (1997) aged a solution recently titrated with a base at 80 °C and found that the conversion from Al₁₃ into AlP₂ was faster than reported by Fu et al. (1991), who studied a "pure" Al₁₃ solution aged at 85 °C. We now know that the AlP₂ species identified by Nazar et al. (Fu et al., 1991; Nazar et al., 1992) corresponds to the Al₃₀ molecule [Al₃₀ = Al₂O₈Al₂₈(OH)₅₆(H₂O)₂₆¹⁸⁺(aq)] (Casey, 2005).

Allouche and Taulelle (2003a) demonstrated that Al monomers [Al(H2O)6]3+ control conversion of the ε-Al₁₃ Keggin species into Al₃₀ by ²⁷Al NMR spectroscopy. Indeed, the presence thereof in solution is crucial to this synthesis. These authors also proposed the chemical pathway illustrated in Figure 5 to explain the thermal isomerization process: ε-Al₁₃ and one aluminum monomer form an intermediate species, ε-Al₁₄, which isomerizes into δ-Al₁₄ at high temperatures. The Al₃₀ cluster then forms via a new dimerization and reaction with two other monomers. Allouche and Taulelle also published a paper (2003b) in which they enhanced the formation of Al₃₀ polycations and other new Keggin isomers by fluorinating the Al₁₃ polycation, partially substituting the di-µ2-OH bridges with fluorides. Barriers have to be overcome in order for ε -Al₁₄ to isomerize into δ -Al₁₄ repulsive electrical energy, thus making the temperature pertinent to the formation of Al₃₀. In this regard, temperatures ranging from 70 to 125 °C have been reported in the literature (Ye et al., 2013). Allouche and Taulelle (2003a) demonstrated that the temperature is not essential to produce Al₃₀. Indeed, these authors reported the involuntary production of Al₃₀ by aging an Al₁₃ solution at 20 °C. However, an increase in temperature accelerates the conversion of Al₁₃ into the Al₃₀ species. One example of this would be heating it for 48 h at 95 °C or 5 h at 125 °C; in addition to the monomer content and heating time, the cumulative Al concentration (AIT) also affects the conversion by facilitating it with lower concentrations and longer aging times (Ye et al., 2013).

Several studies have proposed that δ -Al₁₃ is a necessary unit for polymerization of the ϵ -Al₁₃ cluster into larger polymeric species, including Al₃₀, which is formed by two δ -Al₁₃ species joined via four octahedrally coordinated Al³⁺ cations located in the central region (Abeysinghe et al., 2013a; Abeysinghe et al., 2013b). However, the mechanism for the transformation of Al₁₃ into Al₃₀ and the energy implied in this process is not completely understood due to the complexity of the proposed steps and the fact that intermediate products are difficult to detect (Ye et al., 2013). Additionally, the calculated energy required for the rotation step in the isomerization process is too high to explain the experimental observations (André Ohlin et al., 2014; Oliveri et al., 2016), as shown in the *Supplementary Material*.

The Al₁₃ cluster, which is considered the direct precursor to Al₃₀, is affected by pH, with aggregation occurring at a pH higher than 6 (Allouche and Taulelle, 2003a; Furrer et al., 1992; Mertens et al., 2016; Shafran et al., 2004; Shafran and Perry, 2005), as mentioned in previous sections. Ye et al. (2013) studied the effect of these alkali-induced Al₁₃ aggregates and Al monomers on Al₃₀ formation at high temperature, using ²⁷Al NMR spectroscopy and dynamic light scattering (DLS) measurements. These authors also reported that Al₁₃ aggregates affect Al₃₀ formation, which increases with aging and decreases upon addition of a base. Similarly, Chen et al. (2005) studied the effect of total aluminum concentration on the formation of Al₁₃ and its transformation into Al₃₀ in an aqueous solution. These authors reported that the rate of conversion of Al₁₃ into Al₃₀ increases markedly as the total Al concentration increases and that, when higher than 0.75 mol/dm³, the Al₃₀ cluster content increases continuously, becoming predominant in a fresh polyaluminum solution heated at 95 °C for 12 h.

Shafran et al. (2004, 2005) studied Al speciation at high temperature (90 °C) for a range of hydrolysis ratios and pH levels by potentiometric titration, ²⁷Al NMR spectroscopy, and DLS. The speciation diagrams of Al-ion hydrolysis as a function of pH and hydrolysis ratios are shown in **Figure 6** for several hydrolysis times. **Figure 6** shows that, at a pH of less than 4 (range considered highly acidic), Al

monomers are the majority species, whereas Al dimers appear at pH < 4.3, although their concentration never exceeds 34% of soluble Al species and, thus, they are considered less important. Additionally, these authors found that the quantity of Al₁₃ decreased as the quantity of Al₃₀ increased, reaching a maximum concentration at a pH of around 5.0-5.5 when heated at 90 °C for 12 and 24 h (Shafran et al., 2004). For short periods of hydrolysis (1 and 3 h), Al₁₃ and Al₃₀ are the predominant species in solution at a pH of roughly 4-6, which corresponds to the approximate hydrolysis ratio range $1.6 \le h \le 2.6$. However, at longer hydrolysis times (>6 h), Al₃₀ is the predominant species (Shafran and Perry, 2005).

Sulfate and selenate anions, and various solid-state techniques, have commonly been used to isolate and characterize both Al₁₃ isomers and larger Alcations (Abeysinghe et al., 2013a). Thus, since 1991, the structures of larger Al- $([AI_{26}O_8(OH)_{50}(H_2O)_{20}]^{12+},$ cations such Al_{26} AI_{30} , and AI_{32} as $[Al_{30}O_8(OH)_{56}(H_2O)_{26}]^{18+}$, and $[Al_{32}O_8(OH)_{60}(H_2O)_{28}]^{20+}$ respectively) have been characterized and found to contain two δ-Al₁₃ species. However, the most widely studied cluster has been Al₃₀, which is now well characterized (Abeysinghe et al., 2012; Allouche et al., 2000; Allouche and Taulelle, 2003a; Fu et al., 1991; Oliveri et al., 2016; Phillips et al., 2003; Rowsell and Nazar, 2000; Sun et al., 2011). The importance of temperature as regards both the formation and transformation of Al₁₃ and Al₃₀ species has also been studied. Thus, Chen et al. (2007) used highly concentrated hydrolytic polymeric Al solutions (HPA) to study this effect. These authors reported that, in fresh solutions, the Al₁₃ cluster was the primary component, but that this situation changed when the solution was heated to 95 °C, where the Al₃₀ content increased to become the predominant component after heating at that temperature for 12 h. Due to the relevance of supramolecular chemistry in the field of separation science, some authors have successfully applied this approach to crystallization in the synthesis and structural characterization of Al₁₃ and Al₃₀ Keggintype aluminum clusters in aqueous solutions. To isolate the ε-Al₁₃ cluster, Drljaca et al. (1999) used p-sulfonatocalix[4]arene, whereas Mainicheva et al. (2006) used cucurbit[6]uril to isolate Al₃₀. This latter species was also studied by Abeysinghe et al., (2012) who used 2,6-naphthalene disulfonate (2,6-NDS) to improve crystallization using the supramolecular approach and similarly, crystallized the intermediate species in the solution, reporting δ-Al₁₃, Al₂₆, and Al₃₀. These authors observed that the Al_{26} polycation comprises two $\delta\text{-}Al_{13}$ clusters, which are polymerized via the vertex-sharing of two hydroxyl groups. Al₂₆ is an intermediate product in the transformation of ε -Al₁₃ into larger polynuclear species. **Figure 7** shows the formation mechanism for large polymeric species such as Al₂₆, Al₃₀, and Al_{32\(\text{BS}\)} and illustrates how the ε-Al₁₃ isomer, which is the majority species in solution at a pH of between 3 and 5, is converted into δ-Al₁₃ upon aging or heating. This cluster is linked to additional monomeric species that form the aforementioned larger polymeric structures. The δ-Al₁₃ isomer could be considered to be the building block of larger polymeric species upon heating or aging (Fairley et al., 2012). This molecule also contains a capping sodium atom (see Supplementary Material). This atom appears to stabilize the δ-Al₁₃ cluster shortly before its polymerization into larger aluminum polycations. Removal of the sodium atom results in formation of the Al₂₆ polycation, which is very likely an intermediate product. Higher hydrolysis ratios imply a lower presence of aluminum monomers and dimers in solution, thereby favoring the formation of Al₂₆ rather than Al₃₀ and Al_{32\overline{0}S} species (Abeysinghe et al., 2012).

The complete mechanism for AI_{13} and AI_{30} formation published by Wen et al. (2019a) is presented in **Figure 8**. It can be seen that the step following monomeric species formation at 60 °C in the hydrolysis process (a.1) is the aggregation of these species via the reactions shown in supplementary material to form dimers (a.1) and trimers (a.2), followed by the AI_{13} polycation (a.4). This AI_{13} formation is controlled by the temperature and aging process (OH/AI ratio = 2.4 and a final pH of 3.93). **Figure 8b** represents equation 7, which shows how the ϵ - AI_{13} initially dissolves and subsequently rearranges to the δ - AI_{13} isomer, then how two δ - AI_{13} species can be linked to four monomers to form AI_{30} (pH solution = 4.21) under appropriate conditions such as high temperature (95 °C in an oil bath). The active sites in the AI_{13} structure have been studied, finding similarities between the oxygen atoms in its

structure (Bradley et al., 1993; Casey et al., 2000; Wehrli et al., 1990). Thus, Casey et al. (2000) employed ¹⁷O NMR spectroscopy to study the oxygen exchange rates between the "O" and "OH" sites in Al₁₃ using ¹⁷O-enriched water. These authors found that, of the two inequivalent μ_2 -OH bridges (inter- or intra-trimer) in the ε -Al₁₃, referred to as μ_2 -OH^a and μ_2 -OH^b (see Supplementary Material), μ_2 -OH^a is the most labile hydroxo site. The transformation of ε -Al₁₃ into δ -Al₁₄ is possible because of the lability of the µ₂-OH^a bridges, with the former being capped with Al^{III} monomers (Allouche and Taulelle, 2003b; Wen et al., 2019a). This scheme shows the importance of the monomeric species, in accord to the conclusions published by Allouche and Taulelle (2003a) and Ye et al. (2013). An excess of monomer in solution is necessary for two reasons: 1) they are the basic units that form and connect δ-Keggin-Al₁₄ and form Keggin-Al₃₀, and 2) because they facilitate Al₃₀ formation by increasing the dissolution rate of ε-Keggin-Al₁₃, which promotes the rearrangement and isomerization processes, how was shown by Yang et al. (2010) who investigated the formation mechanism of Al₃₀ using density functional theory. Similarly, Ye et al. (2013) studied the effect of monomeric Al on Al₃₀ formation, as well as the influence of Al₁₃ aggregates, obtained upon alkali addition or aging. These experiments were carried out under high temperatures (95 °C) using characterization techniques such as DLS and ²⁷Al NMR spectroscopy. These authors found that Al₁₃ is also a necessary precursor in the formation of Al₃₀, and that this formation is faster and gives higher yields when Al₁₃ aggregates are formed by aging rather than by titration.

In conclusion, the polymerization of Al₁₃ to Al₃₀ can be achieved by long term aging or additional heating (127 °C for 5 h, or 95 °C for 48 h) of a concentrated Al₁₃ solution (Abeysinghe et al., 2013b; Mertens et al., 2012; Ye et al., 2013). This reaction can be enhanced using an autoclave, thus resulting in a high yield of Al₃₀ (Motalov et al., 2017). Moreover, Al(OH)₃ formation in Al₁₃-containing solutions is not a problem because the formation of the Al₁₃ polymer is significantly faster than the precipitation of Al(OH)₃ phases (Furrer et al., 1992).

4. Acidity and reactivity

As a result of the hydroxide ions and water groups in the structures, both Al₁₃ and Al₃₀ exhibit high reactivity (Mertens, 2011). Thus, both undergo deprotonation over a broad pH range and exhibit a highly stable structure, nanoscale molecular size, a large surface area, and a high surface charge (+7 and +18 respectively) depending on the [OH]/[AI] ratio and pH (Bottero and Bersillon, 1988; Casey, 2005; Furrer et al., 1992; Mertens, 2011; Rakotonarivo et al., 1984; Wu et al., 2016; Ye et al., 2013). However, the particle size of Al₃₀ is nearly twice as large, and its positive net charge is more than 2.5 times higher (Wen et al., 2019a). Al₃₀ is also more acidic than its precursor (ε-Al₁₃) as a result of acidic η-H₂O functional groups, as published by Rustad (2005), who calculated the relative degree of protonation of Al₃₀ and was the first to use molecular dynamics simulations of an acidimetric titration. The deprotonation of AI_{13} starts at pH = 6, as mentioned in preceding sections, and the results obtained for both Al₁₃ and Al₃₀ clusters are similar. One difference between them was observed in the slope of their titration curves, which is steep for ε-Al₁₃ and more gradual for Al₃₀. This indicates that the initial deprotonation events are defined, reaching neutralization when the pH of the solution is approximately 6.7 (pH of zero charge, pH_{PZC}) (Abeysinghe et al., 2013b). Al₃₀ deprotonates in the pH range 4 to 7 (Rustad, 2005), and its high reactivity is due to the large number of OH- and H₂O groups (Mertens, 2011). Rustad (2005) published a theoretical study of the reactivity of Al₃₀ using molecular dynamics simulations and keeping the notation and group numbering used by Casey et al. (2001). A scheme with the oxygen functional groups in Al₃₀ and Al₁₃ is included in **Figure 9**. Al₁₃ contains twelve identical water molecules (η-H₂O) (9a), and two types of μ₂OH bridging groups (between dimeric or trimeric groups) (9b). Similarly, Al₃₀ contains 26 bound terminal water molecules (η-H₂O), which can be divided into five groups depending on their proximity to the central region where both δ-Keggin isomers are linked (9c), 50 doubly bridging (μ₂-OH) ligands, and six triply-bridging (µ₃-OH) ligands (9d) (Casey et al., 2001; Rustad, 2005). Rustad (2005) also reported that the two 4-ηH₂O groups are the most acidic groups in the Al₃₀ molecule and, after deprotonation, form H₃O₂- ligands with the two 3-nH₂O groups. A strong coupling of these sites was also observed.

In 2013 and 2015, Forbes and coworkers published three studies in which they combined theoretical and experimental research. In the first (Abeysinghe et al., 2013b), these authors synthesized and characterized two models of Al₃₀ modified with chelated metal centers (Al^{III} and Zn^{II}), using an iminodiacetic acid (2,6-NDS, IDA) and nitrilotriacetic acid (NTA), respectively (Al₃₂IDA and Zn₂Al₃₂NTA). They found that the reactivity in their experimental models of Al₃₀ accord to the reactive sites predicted by molecular dynamics simulations. In their second paper (Abeysinghe et al., 2013a), they studied the adsorption of Cu2+ on the surface of Keggin-type polyaluminum species (Cu₂Al₃₀-S) crystallized in the presence of disulfonate anions, exploring the adsorption process using density functional theory (DFT). Their results suggested that the reactivity of Al₃₀ toward anions and cations differs, with anions preferring the beltway and cations preferring the caps of the Al₃₀ structure. This study is shown in Figure 10a, where the red surfaces represent regions with higher values of electrostatic potential (highest in the beltway) and the blue surfaces represent regions with lower values (lowest at the caps). In the third paper (Corum et al., 2015), these authors used (TBP)₂Al₃₀-S (TBP = tbutylphosphonate) as a model to study the reactivity of oxyanions with aluminum hydroxide surfaces. They found that the beltway region of Al₃₀ is the most reactive and that preferential binding of phosphate occurs in this region. Al₃₀¹⁶⁺ ([Al₃₀O₈(OH)₅₈(H₂O)₂₄]¹⁶⁺) was used as the initial form in the adsorption modeling using DFT vibrational calculations, finding that Al₃₀¹⁸⁺ ([Al₃₀O₈(OH)₅₆(H₂O)₂₆]¹⁸⁺), which is the molecular formula expected of Al₃₀, is unstable. These findings support the results obtained by Casey and Rustad (2005b), using perchlorate anions (Figure **10b**), who found that the belt region has the highest acidity in Al₃₀ as the highest deprotonation occurs in this region.

5. Al₁₃- and Al₃₀-PILC montmorillonites

Since the 1970s, when the first family of aluminum intercalated clays was reported, PILC have attracted increasing interest from the scientific and industrial

sectors. Several patents have also been granted for such compounds, including three granted to Vaughan, Lussier, and Magee (all assigned to W.R. Grace and Co.), covering the concept of pillaring using specific inorganic polymers (Cañizares et al., 1999; Gil et al., 2000a; Vaughan, 1988; Wen et al., 2019a; Zhu et al., 2017). In 1988, Vaughan published "Pillared clays - historical perspective", in which he described the patents mentioned (Vaughan, 1988). The first report of the structure of ε -Al₁₃ was published in 1960 by Johansson (1960), and the first reports of pillared clays using montmorillonite and Al₁₃ polycations appeared in the period 1977–1981 (Gil et al., 2000a). Since then, several studies that attempt to understand and improve the properties of pillared clays have been published. Various reviews in this field have tried to bring together the extensive published information regarding the synthesis and characterization of PILC, their application in catalysis and adsorption, and the techniques used to characterize them (Baloyi et al., 2018; Ding et al., 2001; Gil et al., 2000a; Gil et al., 2011c; Gil et al., 2008; Kausar et al., 2018; Kloprogge et al., 2005; Kloprogge, 1998; Lazaratou et al., 2020; Ngulube et al., 2017). The majority of studies on PILC involve Al₁₃, although in some cases this is simply an inference due to the experimental conditions utilized. However, under thermal conditions, the Al₁₃ cluster is normally converted into another species. This means that a large proportion of the papers published may actually have used a mixture of Kegginclusters (including Al₃₀) or clusters larger than Al₁₃ as an intercalating solution in which the Al₁₃ species may be only a minor component or even absent (Parker et al., 1997; Smart et al., 2013). The use of Al₃₀ in this field has received little attention even though its characteristics make it an appropriate intercalant that can improve the characteristics of the PILC (Wen et al., 2019a; Zhu et al., 2017).

The characteristics of AI-PILC montmorillonites, including their pore size distributions, surface areas, and surface acidities, can change depending on the preparation procedures and affect their applications, as concluded by several authors such as Mokaya and Jones (1993), Sterte and Otterstedt (1987), Matsuda et al. (1988), Trillo et al. (1993), and Moreno et al. (1997). Similarly, Zuo and Zhou (2008) found a relationship between the pore structure of AI-PILC and the conditions

used in its synthesis. These authors also studied the relationship between the structure of these materials when used as a support in the catalysis of benzene oxidation and their catalytic properties in this process. Gil et al. (2008) reviewed studies related to the control of the microstructure in Al-PILC, including controlling characteristics by adjusting the different parameters in the synthesis process. This review shows how the characteristics of PILC depend on the size, chemical nature, and number of intercalated species, among other aspects, and that modification of the thermal step can affect both the interpillar and interlayer spacings. The type of species formed, which differ in terms of size and net charge, depends on the experimental parameters used. Temperature and pH were two factors mentioned in the "Al₁₃ and Al₃₀ Formation" and "Aluminum Hydrolysis" sections.

Pinnavia et al. (1984) compared the physical and catalytic properties of PILC obtained using natural and synthetic smectite. Cañizares et al. (1999) also compared PILC prepared commercially and naturally (bentonites) with single (Fe, Cr or Zr) and mixed-oxide pillars (those metals and aluminum) and observed larger basal spacings and surface areas in PILC prepared from commercial bentonite than in those prepared from natural bentonite, which may be related to the higher crystallinity of the commercial bentonite used. The incorporation of aluminum into the single oxide enhanced the thermal stability, surface acidity, and methane adsorption. However, the Al/metal ratio used also played an important role by affecting the pillar structure, with higher values making the Keggin structure predominant. Although the basal spacing of the clay is 0.96 nm, its pore size normally ranges from 1.8 to 2.0 nm after insertion of the Keggin-type polycations. Accordingly, the pore size openings in the PILC depend on the size of the polycations (Cañizares et al., 1999); changing the nature and, therefore, the pillar size results in different pore sizes in the PILC, thus making it possible to tune the porosity in such solids (Cool and Vansant, 1998; Maes et al., 1997). The dimensions of the Al₁₃ Keggin structure were proposed in 1988 (Clearfield and Roberts, 1988) and found accord to the interlayer distance in the Al₁₃-PILC montmorillonite (0.8 to 1.0 nm) (Gil et al., 2008).

Butman et al. (2015) reported the structural and textural properties of some PILC obtained by intercalating montmorillonite with large AI (AI₁₃ and AI₃₀) and AI/Ce polyhydroxo complexes. These authors reported an increase in the basal distance (nm) of the montmorillonite and in the surface areas (S_{BET}, m²/g) due to an increase in polycation size from 1.26 (initial) to 1.64 (108 m²/g), 1.88 (125 m²/g), and 2.41 nm (147-154 m²/g) for Al₁₃, Al₃₀, and Al/Ce pillars, respectively. Similarly, Zhu et al. (2017) also published an Al₃₀-PILC montmorillonite synthesis based on the method by Butman et al. (2015), which they considered to be a good attempt at such a synthesis despite the fact that the results published by the latter do not show an obvious enhancement in the properties and that this method also involves a complicated hydrothermal process. For this reason, Zhu et al. (2017) announced theirs was the first successful reported synthesis of an Al₃₀-PILC montmorillonite. These authors used characterization techniques such as X-ray diffraction (XRD), Xray fluorescence (XRF), ²⁷Al NMR spectroscopy, transmission electron microscopy (TEM), and nitrogen adsorption-desorption isotherms, and obtained solids with surface areas (S_{BET}) of 70 (clay mineral), 259 (Al₁₃-PILC), and 311 m²/g (Al₃₀-PILC), which show a clear enhancement in the properties. Furthermore, they reported an increase in the $(Al_2O_3)/(SiO_2)$ ratio from 0.27 (clay) to 0.46 (Al₁₃-PILC) and 0.60 (Al₃₀-PILC). The Al₃₀-PILC formation process published by Zhu et al. (2018) is presented in **Figure 11**, which shows how two adjacent Al₃₀-cluster arrangements in the interlayer of the montmorillonite can create a more sizeable pore. As a future challenge, these authors proposed a change in the orientation of Al₃₀ pillars from horizontal to vertical, thus resulting in PILC with a higher microporosity and larger interlayer spacing, thereby furthering their applications as catalysts and adsorbent materials. Gil and Vicente (2020) considered that a future challenge in the synthesis of PILC could be to explore new structures and find new sources of polycations, including the use of wastes such as industrial metal wastes. Other challenges include their reuse and adaption for specific applications by designing and controlling the porous structure.

6. Use of both Al_{13} and Al_{30} polycations as intercalating agents in PILC

Keggin-type polyaluminum clusters are well-defined species that have been extensively characterized over the last six decades (Abeysinghe et al., 2013a), as discussed previously. These substances are environmentally significant and have been used in various fields. For example, they are considered essential in industries that release large quantities of organic compounds into wastewater, such as the paper industry. They are generally used as flocculants or to form cationic sols for the adsorption of organic compounds in water by adding industrial chemicals such as aluminum chlorohydrate or polyaluminum chloride, which comprise ε-Al₁₃, and Al₃₀ (Bottero and Bersillon, 1988; Casey, 2005; Phillips et al., 2016). The hydrolysis/precipitation behaviors (AIII, AI₁₃, and AI₃₀) and use of AI₃₀ species as a coagulant to enhance turbidity removal have also been studied (Chen et al., 2006; Chen et al., 2009), as has their application in the uptake of heavy metals such as arsenic during water treatment (Mertens, 2011; Mertens et al., 2012; Mertens et al., 2016). In addition to their use as water-clarifying agents, polyaluminum Keggin-type species are also used as pillaring agents in the synthesis of pillared clays (PILC) from layered clays. They are also precursors for heterometallic catalysts upon deprotonation of their amphoteric functional groups, which is related to their ability to adsorb a range of inorganic and organic species on their surfaces (Abeysinghe et al., 2013b; Rowsell and Nazar, 2000).

As mentioned in the Introduction section, the procedure for synthesizing pillared clays can be understood to involve two main steps, (**Figure 2b**): intercalation and calcination. **Figure 12** shows a simple scheme of the general process for synthesizing PILC. The first step involves the preparation of the intercalating solution. One of the most common inorganic intercalating agents for clays is ε-Al₁₃ (Casey, 2005). Indeed, Cool and Vansant (1998), in a chapter of "Synthesis. Molecular Sieves", cited four general methods for the preparation of Al-intercalating solutions to form the Keggin-Al₁₃ polycation, namely two possible hydrolyses of Al^{III} salts, using AlCl₃ and a strong base (NaOH) or with Na₂CO₃, electrolysis of AlCl₃, or

the dissolution of AI metal in HCI. The first method is the most widely used on a laboratory scale, and the latter on an industrial scale, producing concentrated hydrolyzed AI solutions, such as *Locron* (4.6 mol/dm³), which can be used as the intercalating solution if diluted to reach the same concentration, OH/AI ratio, and pH used in the first method (Cool and Vansant, 1998). The use of these commercially available chemicals has been studied in both pillared clays and water treatment, sometimes comparing their use to that of pure AI₁₃ and AI₃₀ solutions (Mertens et al., 2016; Pinnavaia et al., 1984). The use of polyaluminum chloride (PAC) as a coagulant in water treatment is well known, and several studies relating to its coagulation mechanism, such as that by Wu et al. (2016), who compared polyaluminum chlorides with high AI₁₃ and AI₃₀ contents, have been published. Its use to remove contaminants has also been studied by Mertens et al. (2016), who compared arsenic uptake and its bonding sites in a few AI-based sorbents, including commercially available PAC and AI nanoclusters.

As regards its use in the PILC field, Pinnavaia et al. (1984), for example, made a comparison between the use of commercial diluted aluminum chlorohydrate (ACH) and the Al₁₃ solution obtained upon hydrolysis of AlCl₃ with NaOH in the pillaring process. These authors showed that both these systems resulted in pillared products with similar characteristics, such as pore sizes and thermal stability, in addition to catalytic properties. Other researchers, such as Sterte et al. (1987, 1988), Mokaya and Jones (1993), Jones and Purnell (1993), and Polubesova et al. (2000), have also reported the use of ACH solutions in the intercalating process.

Use of the titration method to produce Al₁₃ from a strong base and dissolved Al^{III} salts (normally NaOH and AlCl₃), with aging times of between 24 and 48 h after titration (Mertens et al., 2012). This process has been studied for a broad range of total aluminum concentrations (10⁻⁵ - 1 mol/dm³), using a variety of techniques (Casey, 2005; Fu et al., 1991; Gil et al., 2011b) to study the polymers formed and the PILC obtained using these solutions.

Over the years, several authors have prepared Al₁₃ intercalating solutions by hydrolyzing aluminum chloride by dropwise addition of a NaOH solution. A few of them, such as Lahav et al. (1978), Plee et al. (1987), Pesquera et al. (1991), and Hutson et al. (1999), have studied the effect of varying the [OH-]/[Al3+] molar ratio. Thus, some groups have reported using this method with [OH-]/[Al3+] molar ratios of 2.0 - 2.4 (pH = 4.0 - 4.7), followed by an aging time of 24 - 48 h at 50 - 60 °C (Bradley et al., 1990; Butman et al., 2015; Cool and Vansant, 1998; Gil et al., 2008; Gil et al., 2007; Gil and Montes, 1994c; Guerra et al., 2008; Wen et al., 2019a; Wen et al., 2019b; Wu et al., 2016; Zhu et al., 2018; Zhu et al., 2017; Zuo and Zhou, 2008), whereas others have used higher temperature titrations (80 – 95 °C) at similar [OH- $1/[A|^{3+}]$ ratios (1.5 – 2.5) (Abeysinghe et al., 2013a; Abeysinghe et al., 2013b; Allouche et al., 2000; Allouche and Taulelle, 2003b; Casey, 2005; Corum et al., 2015; Gil and Montes, 1994a, b), some of them subsequently aging the resulting mixtures at room temperature, in some cases for 1 - 2 weeks. Allouche and Taulelle (2003a), for example, confirmed that it is possible to obtain a pure ε -Al₁₃ solution by direct synthesis at an [OH-]/[Al3+] molar ratio of 2.46 with an aluminum concentration of 0.5 mol/dm³ at 100 °C by in situ ²⁷Al NMR spectroscopy. Others have studied the hydrolysis of aluminum chloride using similar $[OH^{-1}/[Al^{3+}]]$ molar ratios (1.6 – 2.2), aging at room temperature for 24 h (Cheng and Yang, 1997; Gil et al., 2000b; Hutson et al., 1999; Molina et al., 1992; Moreno et al., 1997) and 1 - 4 weeks (Shabtai et al., 1984; Tokarz and Shabtai, 1985), or by aging at reflux temperatures in the range of 85 – 100 °C for short periods of between 3 and 48 h (Salerno et al., 2001; Tokarz and Shabtai, 1985; Trillo et al., 1993). Tokarz and Shabtai (1985) reported that PILC obtained under reflux were thermally more stable and also had higher porosity than those obtained at room temperature. This stability is possible because the pillars in these PILC are likely in bigger clusters than Al₁₃ produced at high temperatures (Parker et al., 1997), as explained in the "Al₁₃ and Al₃₀ Formation" section.

Keggin-Al₃₀ intercalating solution can be obtained using the hydrothermal technique, maintaining the Al₁₃-containing solution (obtained by hydrolysis titration or using a commercial solution) at high temperatures (Butman et al., 2015), or using higher temperatures in the aluminum hydrolysis titration procedure. For example,

authors have reported the use of 60 °C for 24 h (Motalov et al., 2017), 80 - 95 °C for 12 – 36 h (Corum et al., 2015; Wen et al., 2019a; Wen et al., 2019b; Wu et al., 2016; Zhu et al., 2018; Zhu et al., 2017), or a higher temperature for a shorter time (127 °C for 5 h) (Butman et al., 2015), stirring in all cases, and using an oil bath (Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2018; Zhu et al., 2017), refluxing (Wu et al., 2016) or a reactor (Butman et al., 2015; Corum et al., 2015). Other authors, such as Vogels et al. (2005) and Feng et al. (2007), have reported the formation of the Al₁₃ polycation by thermal decomposition of urea. Similarly, Sivaiah et al. (2010) reported the use of microwave instead of conventional heating to assist the formation of aluminum polycations/oligomers, especially the Al₁₃ polycation, in both conventional titration and thermal decomposition of urea. These authors used both polycation and oligomer solutions to prepare Al-PILC montmorillonites and found that those prepared using the oligomers formed during the breakdown of urea had a higher micropore volume and surface area. The presence of other Al oligomers higher than Al₁₃ in the thermal decomposition of urea assisted by microwave heating was also observed.

7. Intercalation process

Once the intercalating solution has been prepared, it is used in the intercalation process to facilitate an ion-exchange reaction between the Al-Keggin ions (Al₁₃ or Al₃₀) in the solution and the interlayer ions in the montmorillonite (Na⁺) (Cool and Vansant, 1998), as mentioned in the Introduction section. The intercalating process requires both polycations and the parent clay to have certain properties. For example, the characteristics of the polycations in the solution, including the amount of positive charge and the charge distribution in their surface area, contribute to the exchange process (Wen et al., 2019a). In the case of the clay, the cation exchange capacity (CEC) is important because it is related to the quantity of pillars intercalated. Higher CEC values may mean that the clay charge is not fully compensated, especially if the polycations used have a low charge or are large. Lower CEC values imply that the clay will need only a few pillars to compensate for its charge, which

could signify ineffective intercalation, especially if the polycations used are highly charged or small (Gil et al., 2008). This was demonstrated in the study published by Suzuki et al. (1988, 1989), who compared Al-PILC prepared by varying the CEC of the montmorillonite. Other authors have also studied the pillar density in Al-PILC montmorillonites (Cheng and Yang, 1997; Gil et al., 2000b; Hutson et al., 1999; Suzuki et al., 1988; Suzuki and Mori, 1989). For example, Cheng and Yang (1997) and Hutson et al. (1999) compared the Al-PILC prepared using two montmorillonites with very different CECs (0.70 and 1.40 mequiv/g) and found that both PILC had almost the same interlayer spacings but differed in terms of pillar density. This was higher in the PILC with a higher CEC in the raw clay, which resulted in a lower microporosity in this PILC. A moderate CEC is therefore required, and smectites or TOT are ideal hosts, as mentioned in the Introduction section (Cool and Vansant, 1998).

The surface acidity of clay minerals is important in several processes and applications, including their use in heterogeneous catalysis (Lambert and Poncelet, 1997; Wen et al., 2019b). The surface acidity of montmorillonite, a TOT clay, arises due to its ability to donate protons, which is determined by the oxygen atoms of the Si tetrahedron in its basal structure. **Figure 13** shows possible acid sites in this clay, which has both Lewis and Brønsted acid sites (Wen et al., 2019b). Salerno et al. (2001) found that acidity is higher in PILC, and the strength and number of the acid sites formed in this process depend primarily on the Al/Clay ratio. PILC have both Lewis and Brønsted acid sites, with the ratio between them depending on the type of clay (Cool and Vansant, 1998). The relationship between the pillars and the change in surface acidic properties of the montmorillonite has also been studied (Bradley and Kydd, 1993; Casey, 2005; Wen et al., 2019b). Thus, Bradley and Kidd (1993) compared the acidic character of montmorillonite after pillaring with various tridecameric cations, including Al₁₃ and GaAl₁₂, and found that the Lewis and Brønsted acidic sites in the resulting PILC depend on the pillars. These pillars affect the acidity mainly because their introduction into clays increases the interlayer spacing, microporosity, and surface areas, thereby exposing the acid sites in the

phyllosilicate sheet (Bradley and Kydd, 1993; Casey, 2005). After the intercalation process, hydrolysis of the Keggin ions continues to a degree that depends on the host type (Cool and Vansant, 1998).

To try to understand the behavior of Keggin polycations during the formation of Al-PILC, and how this is related to their improved properties, Wen et al. (2019 a,b) performed two studies of the relationship between the pillars, the surface acidity of the clay, and the formation and transformation of these pillars during the entire process (from hydrolysis to calcination). In the first study (Wen et al., 2019b), they demonstrated the effect of the pillars on the surface acidity of montmorillonite and their relationship to the Al/Si ratio by comparing the catalytic oxidation of toluene using montmorillonite and Al₁₃- and Al₃₀-PILC montmorillonites. To study the impact on surface acidity when the montmorillonite was pillared using Al₁₃ and Al₃₀-Keggin polycations, they used NH₃-TPD (NH₃ temperature-programmed desorption) and NH₃-adsorbed DR-FTIR (NH₃-adsorbed diffuse reflectance Fourier transform infrared). This study showed that it is possible to propose arrangement models for these polycations in the interlayer space of the clay mineral (Figure 14) based on the results obtained, including the results of powder XRD and the chemical composition and structural parameters of the clay mineral. These authors also found that both Al₁₃- and Al₃₀-PILC showed an increase in surface acid sites, mainly Brønsted acid sites (both strong and weak), although the number of these sites depended on the Al/Si ratio. A high catalytic activity was reported for Al₃₀-PILC at a much lower temperature than for the others. Particle sizes of approximately 0.9 x 0.9×0.9 (Keggin-Al₁₃ polycation) and $0.9 \times 1.2 \times 2.0$ nm³ (Keggin-Al₃₀ polycation) were considered for the models in Figure 14. Both these systems involve electrostatic interactions with the layers of the clay and are configured differently in this interlayer region. This difference in configuration explains why both Al-PILC differ in their surface area and in the number of micro- and mesopores (Wen et al., 2019b). The symmetrical structure of Al₁₃ is similar to a sphere, therefore the surface charges are distributed evenly and arranged in the interlayer region in a nonspecific manner. The specific "flat-lying" arrangement of the Keggin-Al₃₀ polycation in the

interlayer region of montmorillonite is a result of its extremely high positive charge and the charge distribution on its surface, with the highest density of positive charges being in the "belt" of the structure, as discussed in the "Acidity and Reactivity" section. This, in addition to its large size compared to Keggin-Al₁₃, results in a PILC with a larger porosity and a larger surface area (Wen et al., 2019a). In the second work, Wen et al. (2019a) studied the formation, transformation, and properties of Al-Keggin structures during the stages, including the hydrolysis stage, the intercalating stage, and the calcination stage, focusing on the Keggin-Al₃₀ polycation. These authors used various characterization techniques, such as field-emission scanning electron microscopy, thermogravimetric analyses, and X-ray diffraction, and later published the schematic representation of **Figure 8** as shown in the "Al₁₃ and Al₃₀ Formation" section. They concluded that, during the formation of Keggin-Al₃₀, the excess of monomeric Al species at high temperature was an important factor, in agreement with other authors, as discussed in that section.

Experimentally, one important parameter during intercalation is the Al/clay ratio used, although the quantity of pillars intercalated depends on the charge of the pillars, the presence of other charged species in the solution, and the CEC of the clay (Cool and Vansant, 1998), as mentioned previously. Over the past few years, several authors have studied the implications of varying the aluminum/clay ratios (Gil and Montes, 1994a; Gil et al., 2007; Molina et al., 1992; Pesquera et al., 1991; Salerno et al., 2001; Stacey, 1988; Tokarz and Shabtai, 1985), and others have used Al/Clay (mmol/g) ratios of 2.0 (Maes et al., 1997; Shabtai et al., 1984), 3.0 (Butman et al., 2015), 4.0 (Matsuda et al., 1988; Matsuda et al., 1987; Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2018; Zhu et al., 2017), 5.0 (Gil et al., 2000b), 7.0 (Moreno et al., 1997), 10.0 (Cheng and Yang, 1997; Hutson et al., 1999), and 20.0 (Zuo and Zhou, 2008) in their studies. Gil et al. (2007), for example, used nitrogen and carbon dioxide adsorption methods to study the structural properties of Al-PILC montmorillonites obtained upon varying the Al/clay ratio (10, 30, 90, and 150 mmol/dm³ Al^{III}/g clay) and calcination temperature (200, 400 and 500 °C). These authors reported that an increase in both the Al/clay ratio and treatment temperature

reduced the surface area and micropore volume of the clays. They also found that a lower Al/clay ratio (mmol/dm³ Al^{III}/g clay) was sufficient to achieve successful intercalation.

For the intercalation of both Al₁₃ and Al₃₀ clusters into montmorillonite, two methods comprising the use of the clay mineral as a powder or use as an aqueous dispersion have been reported in the literature. However, PILC prepared using the clay mineral as a powder are much more affordable and produce highly porous, homogeneously intercalated clays (Cool and Vansant, 1998). Some of the parameters reported for each intercalation method and each polycation (Al₁₃ and Al₃₀) are presented below. Thus, for the Al₁₃ polycation, studies involving dropwise addition of the intercalating solution to an aqueous suspension (0.002 - 2.5%) of the montmorillonite with stirring (Bradley et al., 1990; Butman et al., 2015; Cheng and Yang, 1997; Gil and Montes, 1994a; Gil et al., 2007; Hutson et al., 1999; Lahav et al., 1978; Moreno et al., 1997; Pesquera et al., 1991; Plee et al., 1987; Salerno et al., 2001; Shabtai et al., 1984; Stacey, 1988; Suzuki et al., 1988; Tokarz and Shabtai, 1985; Zuo and Zhou, 2008), at room temperature for 12, 24, or 48 h (Cheng and Yang, 1997; Gil and Montes, 1994a, b; Gil et al., 2007; Hutson et al., 1999; Molina et al., 1992; Moreno et al., 1997; Suzuki et al., 1988), or, in some cases, at higher temperatures such as 60 °C for 3 h (Zuo and Zhou, 2008), or 80 °C for 2 h (Butman et al., 2015), have been reported. Some of these mixtures were aged for 12 h (Butman et al., 2015) or 24 h (Pesquera et al., 1991) after the addition had finished. When using montmorillonite as a powder, the addition thereof to intercalating solutions of Al₁₃ with continuous stirring has been reported at room temperature for 24 h (Gil et al., 2000b), at 70 °C for 1 h (Matsuda et al., 1988; Matsuda et al., 1987), and at 60 °C for 24 h, and when using the latter method, the mixture was aged for 24 h at the same temperature (Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2018; Zhu et al., 2017). Various authors have explored alternatives to improve the synthesis method by making it a shorter and less tedious process. Thus, Fetter et al. (1996 and 1997) and Sampieri et al. (2004) compared the conventional intercalation process with the use of a version involving microwave irradiation and found that the use thereof has advantages, such as making the pillaring process possible in less than 15 minutes, the elimination of organic impurities from the clay, and the possibility of using both concentrated aluminum solutions and clay suspensions in the intercalation process.

In the case of the Al_{30} polycation, the use of a 1 wt.% aqueous suspension of montmorillonite has been reported, adding the intercalating solution to it dropwise at 80 °C for 2 h, followed by aging for 12 h at room temperature (Butman et al., 2015). The dispersion of montmorillonite as a powder into the intercalating solutions of Al_{30} has been reported at 95 °C for 24 h, with an aging time of 24 h at the same temperature (Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2018; Zhu et al., 2017). In both cases, the procedures were carried out with continuous stirring.

8. Washing, drying, and calcination steps

The intercalation process is followed by the separation of the intercalated clay and the solution, which includes two important steps, namely washing and drying the intercalated clay. These steps precede the calcination required to obtain the pillared clay and both of them seem to improve the quality of the PILC. The washing step promotes an equal dispensation of pillars between the layers and forms larger interlayer spacings. Indeed, differences exist between the Al-PILC if they are unwashed (1.2 nm) or washed (1.8 nm) (Cool and Vansant, 1998), whereas the drying step can affect the pore size and its uniformity (Gil et al., 2008; Salerno et al., 2001). Cooling to room temperature is necessary in some cases. Both filtration (Cheng and Yang, 1997; Hutson et al., 1999; Matsuda et al., 1988; Matsuda et al., 1987; Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2018; Zhu et al., 2017; Zuo and Zhou, 2008), and centrifugation (Butman et al., 2015; Gil and Montes, 1994a, b; Gil et al., 2007; Gil et al., 2000b; Stacey, 1988; Suzuki et al., 1988; Tokarz and Shabtai, 1985) have been used in the separation process. In both cases, the washing process is repeated on the solid obtained, using distilled water to remove both chloride and excess interclating solution. Finally, to dry the samples, there are

reports of slow drying in an oven at temperatures of 50 – 60 °C (Butman et al., 2015; Gil et al., 2007; Gil et al., 2000b; Suzuki et al., 1988), 80 °C (Salerno et al., 2001), and 20 - 100 °C, (Gil and Montes, 1994a, b; Hutson et al., 1999; Matsuda et al., 1988; Matsuda et al., 1987; Stacey, 1988; Trillo et al., 1993; Zuo and Zhou, 2008), in some cases overnight. Quick-drying techniques such as freeze-drying have also been reported (Moreno et al., 1997; Pesquera et al., 1991; Plee et al., 1987; Tokarz and Shabtai, 1985; Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2018; Zhu et al., 2017). However, to obtain a well-ordered PILC, Trillo et al. (1993) recommended air-drying the sample instead of freeze-drying because the products obtained by freeze-drying showed meso- and macroporous structures. The use of air to dry the intercalated solids favors crystallinity and microporosity because this slow drying method allows the layers to fix themselves in a parallel and ordered manner (faceto-face stacking), whereas in quick drying methods, like freeze-drying, the aggregates form in a random manner (Cool and Vansant, 1998). The washing and drying steps are followed by calcination of the intercalated solid. This heating process is required to produce stable pillared clay. During this process, dehydration and dehydroxylation of the Al-polyoxocations occur, thus converting these precursors into neutral and rigid alumina oxide pillars (Casey, 2005; Cool and Vansant, 1998). The PILC obtained are permanently microporous and will not be affected by phenomena like expansion or hydrolysis, although the protons remaining in their interlayer regions remain candidates for ion exchange (Cool and Vansant, 1998).

The preceding sections noted that, among the experimental parameters that can be adjusted in the synthesis process to control the characteristics and microstructure in Al-PILC, the calcination step has shown to have a marked effect on the texture of the Al-PILC montmorillonites, as shown by Gil et al. (1994b, 2007, 2008). In this regard, Zhu et al. (2018) used techniques such as thermogravimetric analysis, N₂ adsorption-desorption, and high-temperature X-ray diffraction to compare the thermal stability of montmorillonites, both natural and pillared using both Al₁₃-, and Al₃₀-Keggin polycations. These authors found that both Al-PILC (Al₁₃

and Al₃₀) exhibit improved thermal stability, and proved that Al₃₀-PILC is the most thermally stable of them. Their results showed that while the layers of the natural clay collapsed at 200 °C, the layers of Al₃₀-PILC remained unchanged, even at a calcination temperature of 800 °C.

The Al_{30} cluster is more compact than the Al_{13} cluster due to the smaller O-Al-O bond angles and shorter η -OH₂ and μ -OH bond distances, thus giving it higher thermal stability. As a result, the Al_{30} -PILC montmorillonite structure is stable up to 800 °C (Wen et al., 2019a). The use of both Al_{13} and Al_{30} polycations in the pillaring process improves the thermal stability of montmorillonite due to the formation of Al oxides with high thermal stability as pillars in the interlayer region. This provides protection and allows retention of the layered structure as micro/mesoporous structures at high temperatures (Wen et al., 2019a; Zhu et al., 2018). In the calcination step, both Keggin polycations (Al_{13} and Al_{30}) lose water molecules as the temperature increases, the first being the η -OH₂ groups due to the relative weakness of the Al-OH₂ bond (Wen et al., 2019a).

Motalov et al. (2017) used mass spectrometry to compare the thermal emission of alkali metal ions (Li⁺, Na⁺, K⁺, Rb⁺, and Cs⁺) in Al₃₀-pillared montmorillonite and its natural form (from 500 to 660 °C in both cases). These authors observed an emission anomaly in the temperature range 532 – 559 °C, which they assigned to the chemical transformations of the pillars and concurrent reactions with the alkali metal ions.

Al₁₃-pillared montmorillonites have been reported as having been obtained after calcination in a furnace at 300 °C for 2-3 h (Butman et al., 2015; Plee et al., 1987; Wen et al., 2019b; Zhu et al., 2017), 350 °C for 12 h (Cheng and Yang, 1997), 400 °C for 2-4 h (Maes et al., 1997; Matsuda et al., 1987; Molina et al., 1992), and 500 °C for 1-4 h (Gil et al., 2000b; Salerno et al., 2001; Suzuki et al., 1988; Suzuki and Mori, 1989; Zuo and Zhou, 2008). To obtain information about the thermal stability of these Al₁₃-PILC montmorillonites, and to study the effect of calcination

temperature on their textural properties, several studies have evaluated several calcination temperatures, such as $200-600\,^{\circ}\text{C}$ for $3-4\,\text{h}$ (Gil and Montes, 1994b; Gil et al., 2007; Moreno et al., 1997; Pesquera et al., 1991; Stacey, 1988), $300-700\,^{\circ}\text{C}$ for $4-6\,\text{h}$ (Matsuda et al., 1988; Trillo et al., 1993), $400\,\text{and}\,600\,^{\circ}\text{C}$ for $12\,\text{h}$ (Hutson et al., 1999), and $200-1100\,^{\circ}\text{C}$ (Wen et al., 2019a; Zhu et al., 2018). Al_{30-pillared} montmorillonites can be obtained after calcination at $300\,^{\circ}\text{C}$ for $2-3\,\text{h}$ (Butman et al., 2015; Wen et al., 2019a; Wen et al., 2019b; Zhu et al., 2017), $350\,^{\circ}\text{C}$ for $3\,\text{h}$ (Motalov et al., 2017), at higher temperatures such as $800\,^{\circ}\text{C}$ (Wen et al., 2019a), and at $200-1100\,^{\circ}\text{C}$ (Zhu et al., 2018). Other authors, such as de Andrés et al. (de Andrés et al., 1999), have studied the use of microwave irradiation instead of the conventional calcination step in aluminum pillared montmorillonite synthesis. Samples obtained in this way were intermediate between the intercalated clays and the PILC obtained using the standard calcination treatment.

9. Conclusions

The results summarized in this review show the great interest that has existed over the years in understanding both formations, namely Al₁₃ and Al₃₀ polycations, and Al₁₃- and Al₃₀-PILC montmorillonites. Some of the studies discussed herein show the effect of total aluminum and aluminum monomer concentrations on the process for synthesizing intercalating solutions. The importance of the experimental parameters used in both synthetic processes mentioned (intercalating solution and Al-PILC) is reflected in the majority of the manuscripts related to Al-PILC in this review. The results clearly illustrate researchers' attempts to propose several conditions for these processes to achieve the successful synthesis of both Al-PILC using Al₁₃ and Al₃₀ polycations as intercalant solutions while guaranteeing reproducible characteristics in the solids obtained. Some authors have also proposed future challenges in this field, which are mainly related to producing Al-PILC with higher microporosity and larger interlayer spacings, thereby further expanding their role and development as catalysts and adsorbent materials.

From the results included in this work, there are still many interesting areas to be explored for applications of PILC materials. The porous structure can be designed, controlled, and adapted to desired applications as the selective adsorption of pollutants, the synthesis of more selective catalysts for wastewater applications, and the development of biocatalysts by the immobilization of enzymes and microorganisms.

Declaration of Competing Interest

No

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CAPTIONS

- **Figure 1.** Schematic structure of the TOT montmorillonite (Reproduced with permission from Liu and Zhang, 2014).
- **Figure 2. (a)** Comparison of the hydration and dehydration behavior of a natural clay mineral and a pillared clay. **(b)** Pillaring process (Reproduced with permission from Cool and Vansant, 1998).
- **Figure 3.** The distribution curve of the different Al^{III} species present in the AlCl₃ solution hydrolyzed with NaOH, depending on the pH and OH/Al ratio. Published by Cool and Vansant (Reproduced with permission from Cool and Vansant, 1998), corresponding to the NMR and potentiometric titration results published by Bottero et al. (Reproduced with permission from Bottero et al, 1980, 1982).
- **Figure 4.** Polyhedral representation: **(a)** Al₁₃; **(b)** AlP₁, **(c)** a possible "unsaturated" structure for AlP₂, **(d)** a "saturated" structure for AlP₂ (Reproduced with permission from Fu et al, 1991).
- **Figure 5.** Proposed formation of Al_{30} initiated by aluminum monomers with ϵ - Al_{13} as precursor species (Reproduced with permission from Allouche and Taulelle, 2003).
- **Figure 6.** Al-ion hydrolysis speciation diagram at 90 °C (Reproduced with permission from Shafran et al., 2004).
- **Figure 7.** The formation mechanism of the large polymeric species Al₂₆, Al₃₀, and Al_{32\(\text{\mathbb{I}}\)S (Reproduced with permission from Abeysinghe et al, 2012).}
- **Figure 8.** The formation process of the polycations Keggin-Al₁₃ (a) and Keggin-Al₃₀ (b) (Reproduced with permission from Wen et al, 2019a).

Figure 9. Surface functional groups: **(a, c)** Bound water molecules (η -H₂O) in Al₁₃ and Al₃₀, respectively; **(b, d)** bridging hydroxide ions in Al₁₃ and Al₃₀ respectively (Reproduced with permission from Rustad, 2005).

Figure 10. (a) The surface charge density of Al_{30}^{16+} (Reproduced with permission from Abeysinghe et al, 2013). (b) Outer-sphere water molecule (top) and anion (bottom) distribution around an Al_{30} molecule in the skeletal form (Reproduced with permission from Casey et al., 2005).

Figure 11. Representation of the synthesis of Al₃₀ (a) and Al₃₀-PILC montmorillonite (b) (Reproduced with permission from Zhu et al., 2017).

Figure 12. Pillared Interlayered Clays: general preparation procedure.

Figure 13. Possible acid sites in the montmorillonite (Reproduced with permission from Wen et al, 2019b).

Figure 14. Al-polycation arrangement models in montmorillonite. Top view of Keggin polycations Al_{13} (**A**) and Al_{30} (**B**), and three-dimensional view of Keggin polycations Al_{13} (**C**) and Al_{30} (**D**) (Reproduced with permission from Wen et al, 2019b).

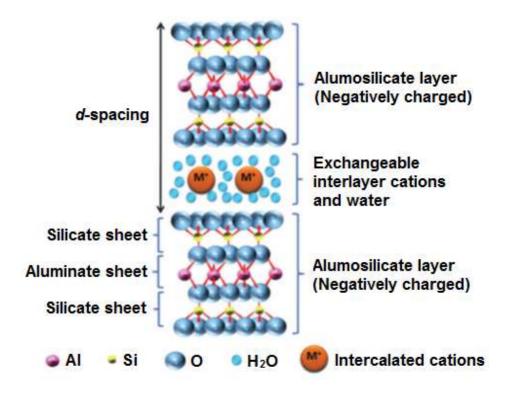


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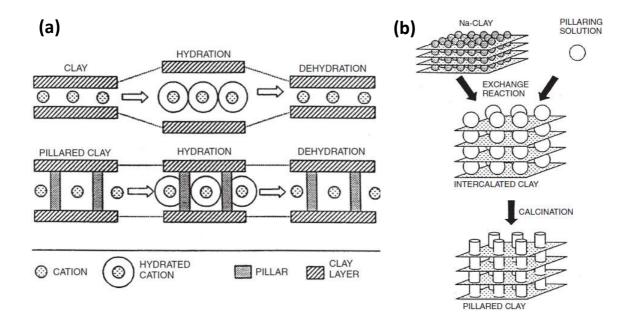


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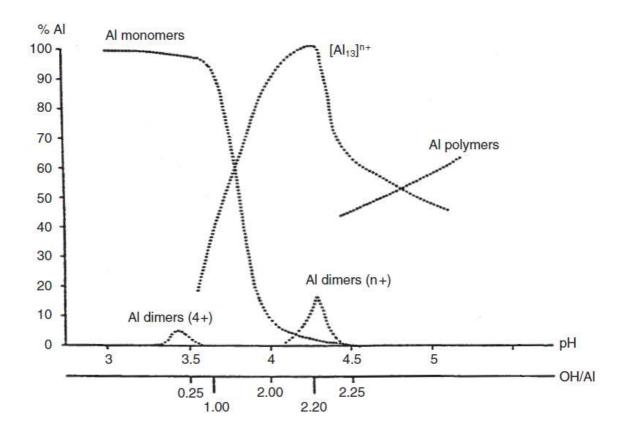


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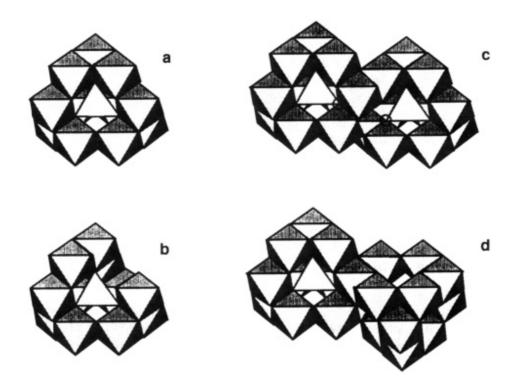


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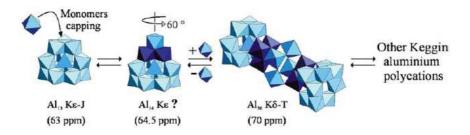


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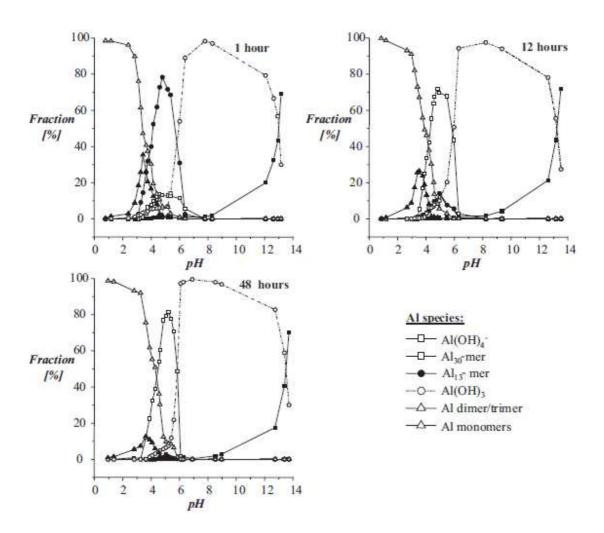


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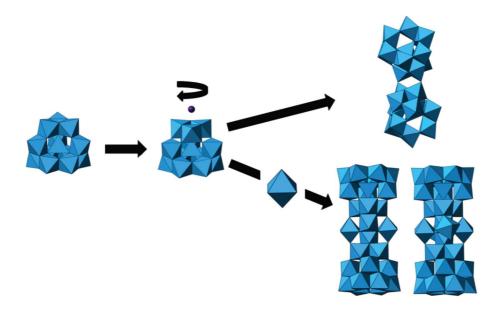


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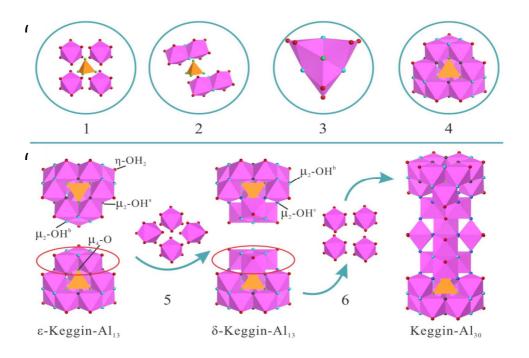


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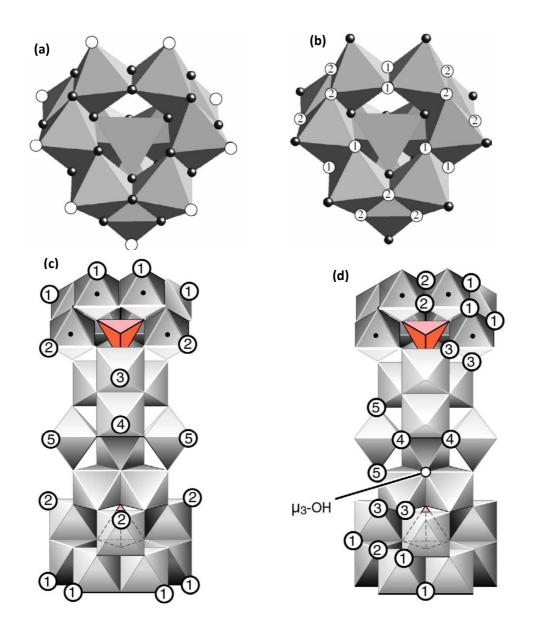


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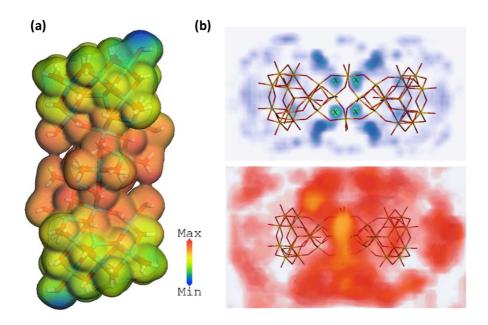


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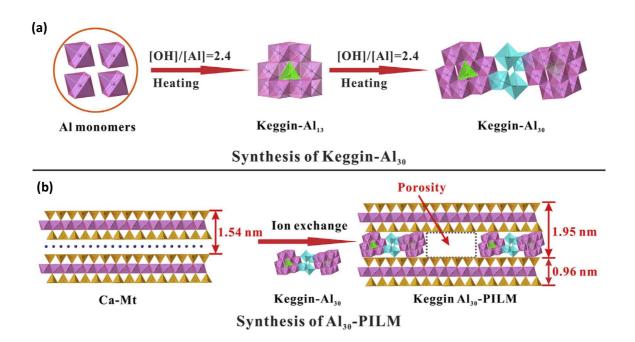


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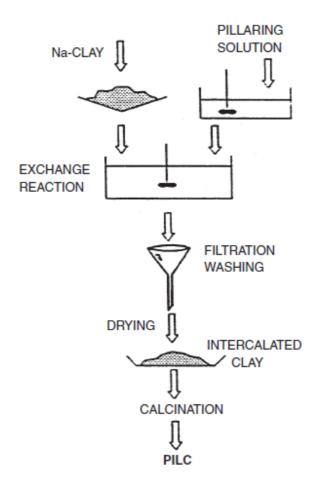


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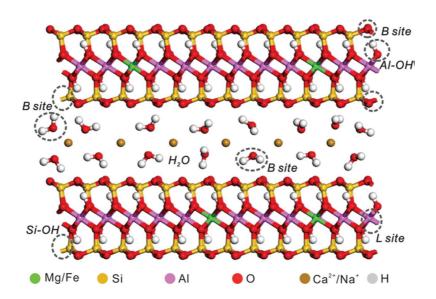


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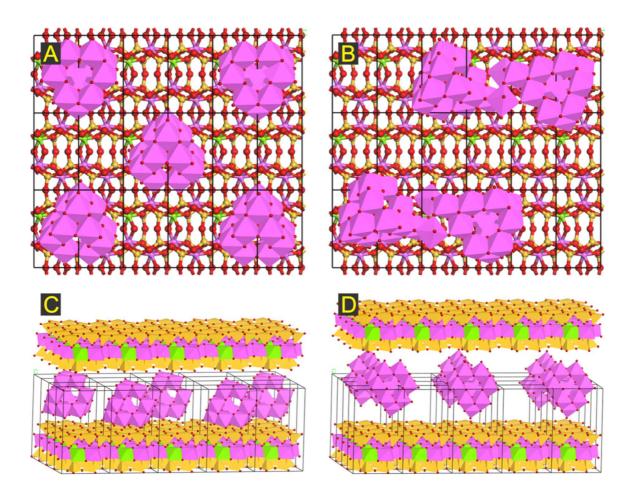


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